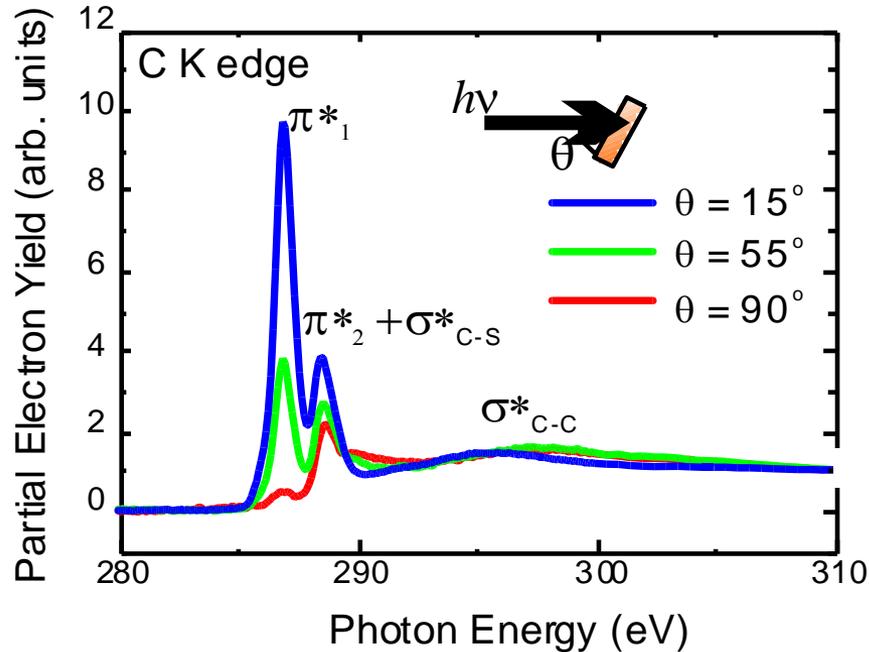


# Soft X-ray Absorption Spectroscopy

Kenta Amemiya (KEK-PF)

1. Advantages and Disadvantages of  
Soft X-ray Absorption Spectroscopy (SXAS)
2. SXAS studies on Surface and Thin films
3. Novel SXAS Techniques
  - 3-1. Depth-resolved XAS
  - 3-2. Wavelength-dispersive XAS

# Soft X-ray Absorption Spectroscopy (~100-4000 eV)



In the Soft X-ray region,

1. Vacuum condition is normally required. (NOT ultra-high vacuum)  
Special sample cell or He atmosphere is available for ambient pressure.
2. Surface sensitive  
several nm for electron yield,  $\sim 0.1 \mu\text{m}$  for fluorescence yield

## 1. Element Selectivity

<- Core-hole excitation (1s, 2p...)  
(C: 290 eV, N: 400 eV, O: 530 eV...)

## 2. Information on Chemical Species

<- Characteristic peaks ( $\pi^*$ ,  $\sigma^*$ ...)

## 3. Structural information (bond length...)

EXAFS (Extended X-ray Absorption Fine Structure)

## 4. Information on Anisotropy

<- Linear polarization  
(molecular orientation, lattice distortion)

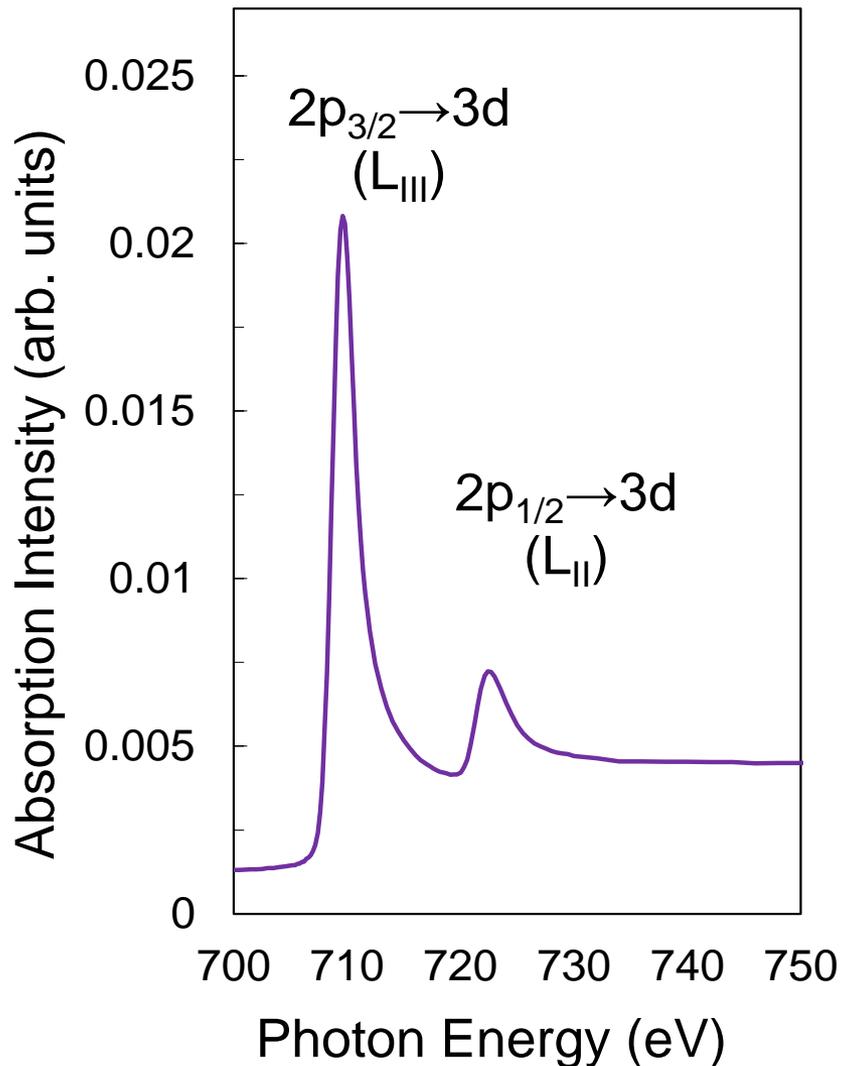
## 5. Magnetic information

<- Circular polarization

## 6. High Sensitivity

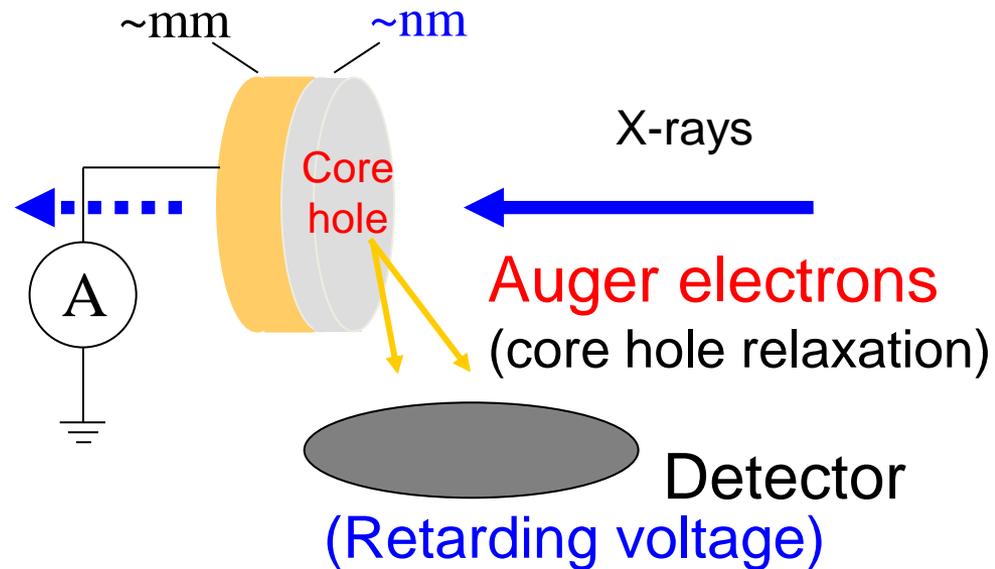
# XAS Measurement in the Soft X-ray Region

3 ML Fe / Cu(100) **Fe L-edge XAS**



How can we measure

**X-ray absorption** spectrum ?



**Electron yield** XAS

Total electron yield (TEY)

Partial electron yield (PEY)

cf. **Fluorescence yield** (FY)

# Advantages and Disadvantages of SXAS

## Short Penetration Length

Transmission mode can be available only for very thin samples with very thin or no substrate.

Electron yield mode is usually adopted because of high efficiency.

Special care is necessary for insulators (powders might be OK).

Fluorescence yield efficiency is very small for light elements.

<1 % for C, N, O

Be careful for the self absorption (saturation) effect.

Samples should be usually kept in vacuum (NOT ultra-high vacuum).

Some attempts have been made to realize ambient-pressure or liquid-state measurements.

## Surface Sensitive

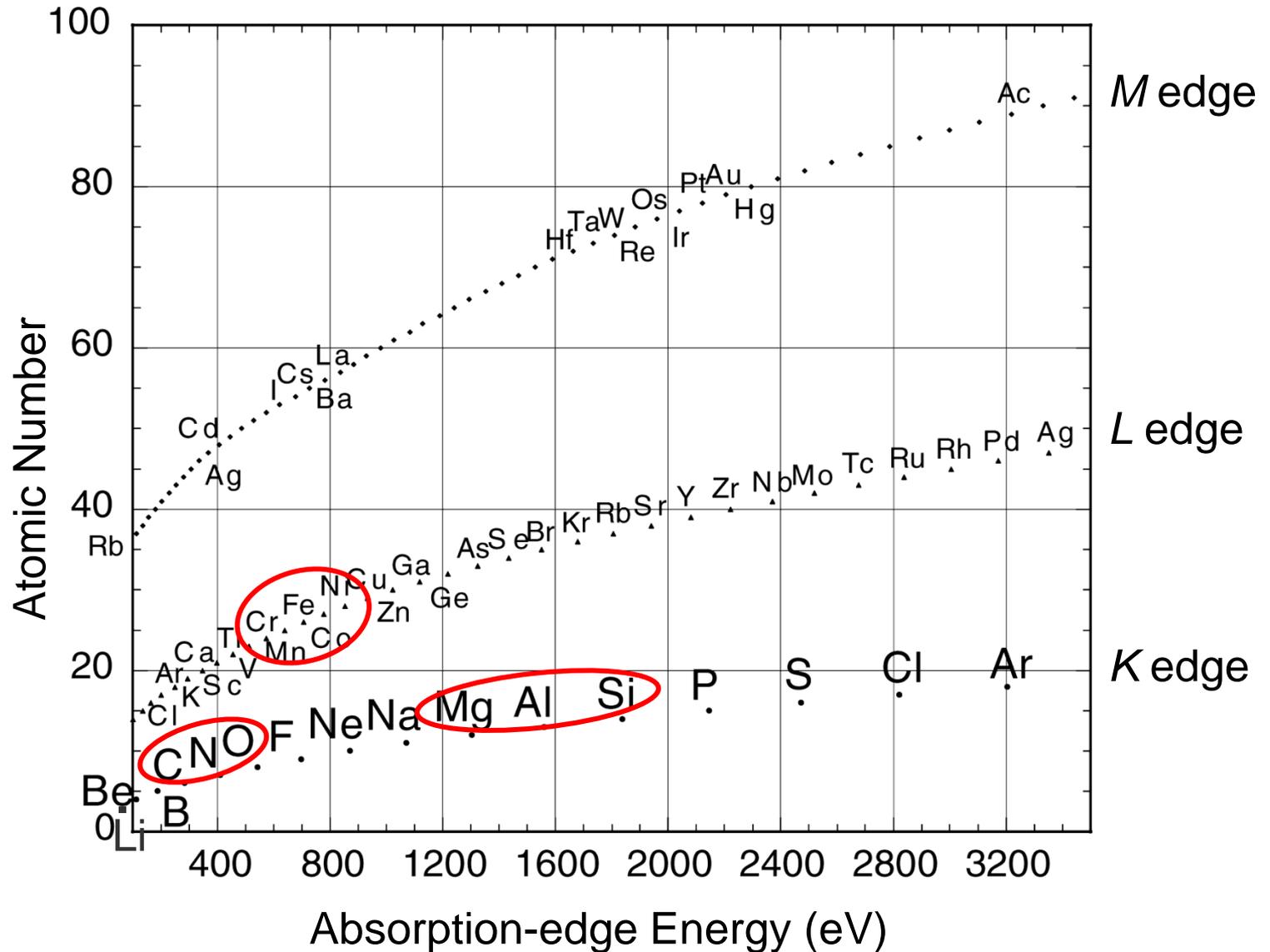
Sub-monolayer samples can be investigated.

Bulk information is hardly obtained, especially in the electron yield mode.

## Sensitive to Electronic and Magnetic States of light elements

Valence electrons can be directly investigated by  $1s \rightarrow 2p$  excitation of C, N, O,... and  $2p \rightarrow 3d$  excitation of 3d transition metals.

# Absorption Edges in the Soft X-ray Region



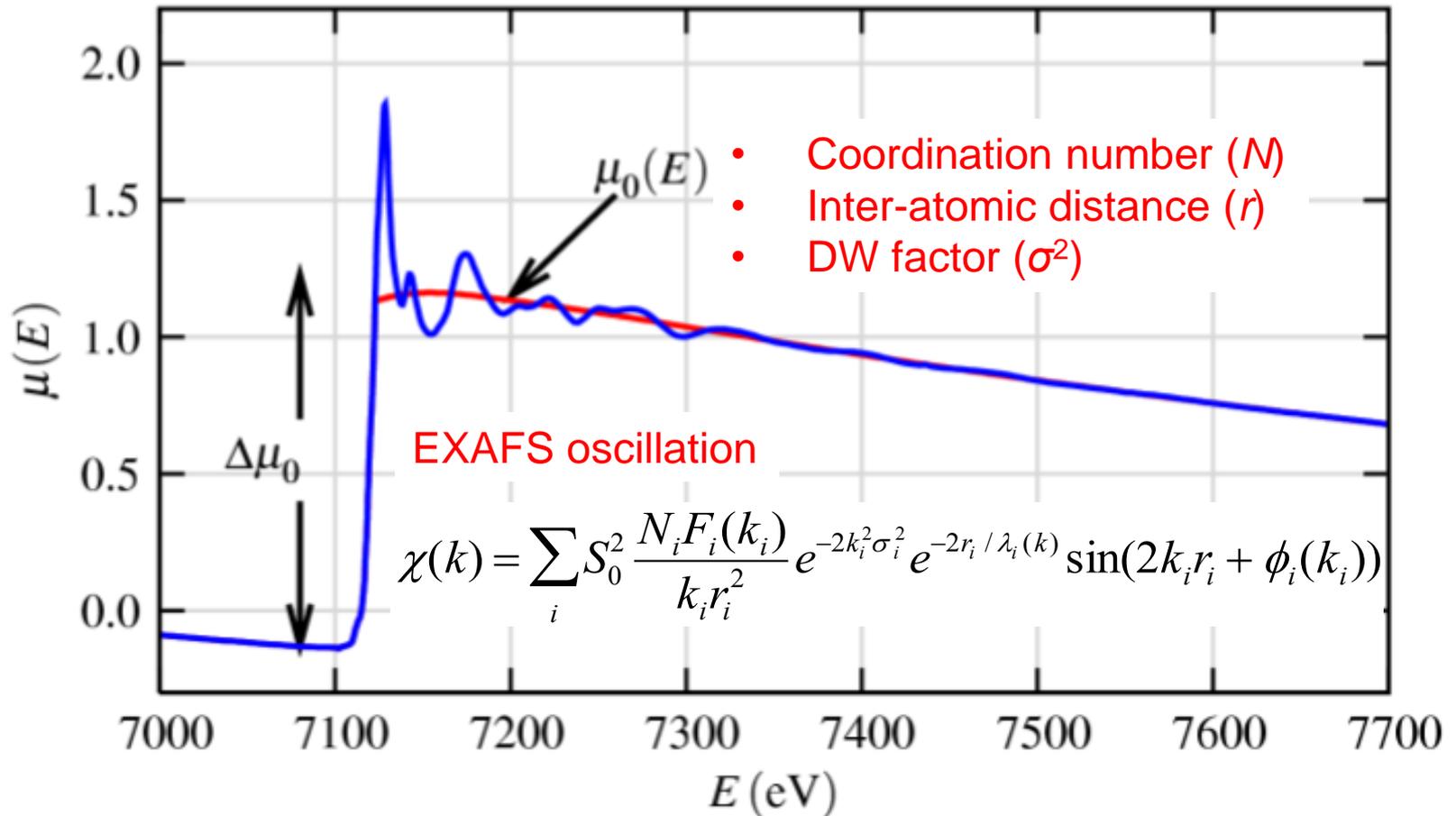
1. Advantages and Disadvantages of  
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# Determination of Atomic Structure

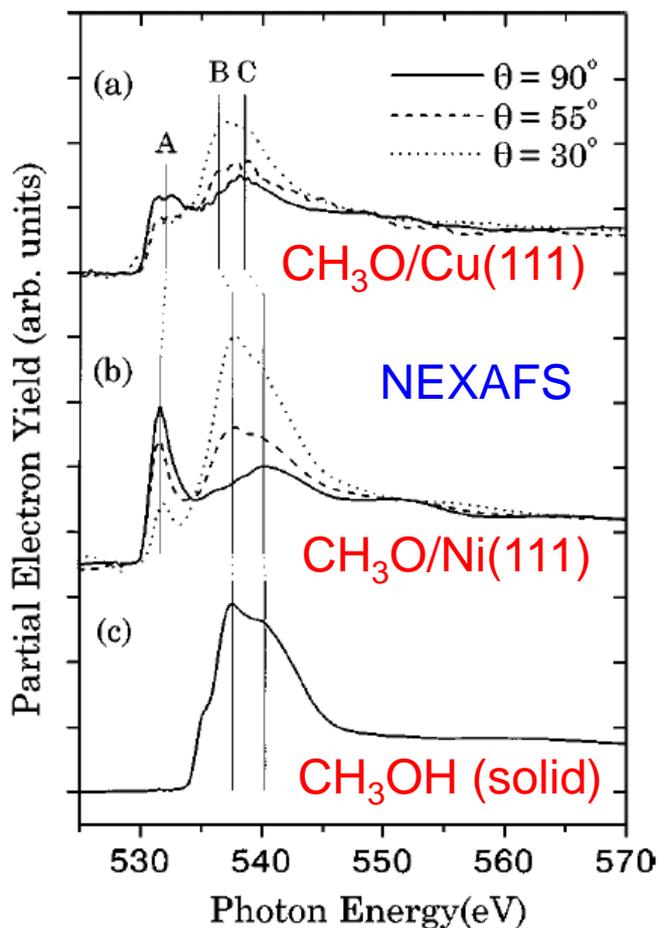
## Extended X-ray Absorption Fine Structure (EXAFS)



Fe  $K$ -edge XAFS spectrum  $\mu(E)$  of FeO

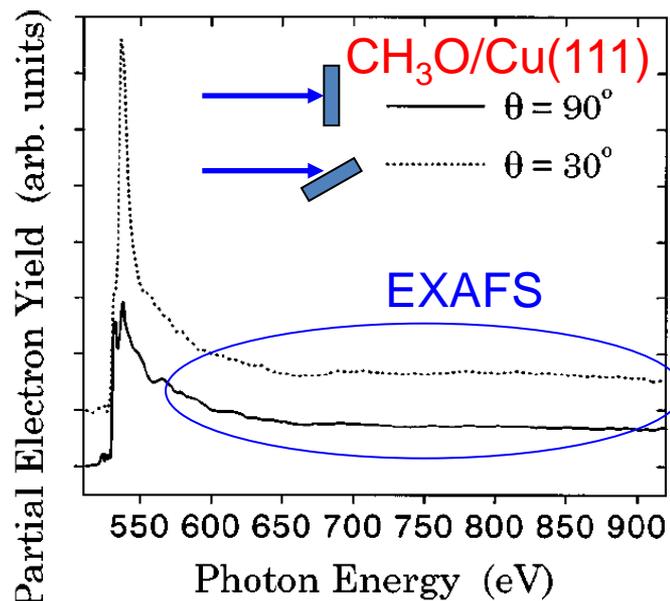
# Determination of Atomic Structure

Amemiya et al., Phys. Rev. B **59**, (1999) 2307.

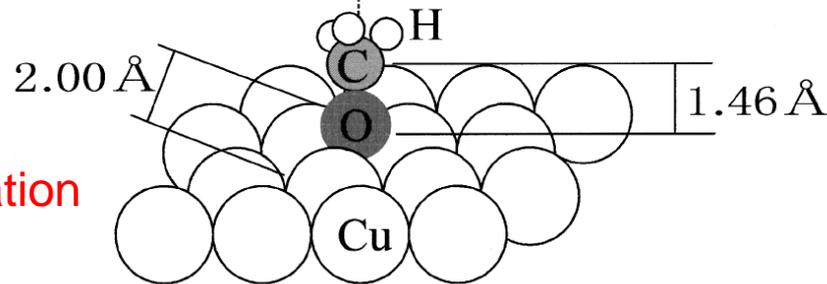


Peak B ( $1s \rightarrow \sigma_{\text{CO}}^*$ ) -> **C-O bond length**  
 Angle ( $\theta$ ) dependence -> **molecular orientation**

Application to surface molecule ( $\text{CH}_3\text{O}$ )

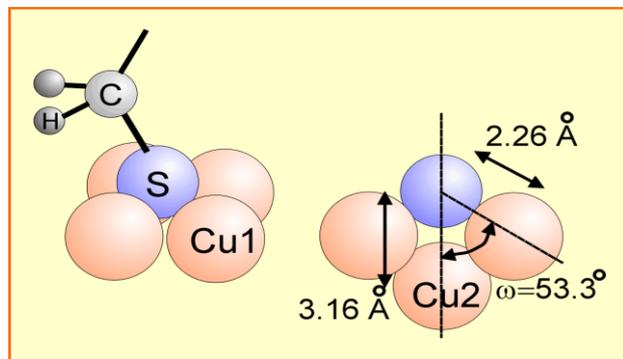
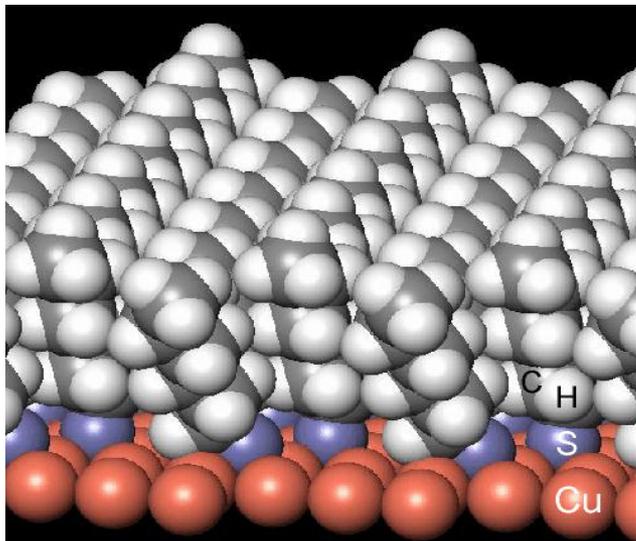


Oscillation period -> **O-Cu bond length**  
 Angle ( $\theta$ ) dependence -> **bond angle**  
**adsorption site**

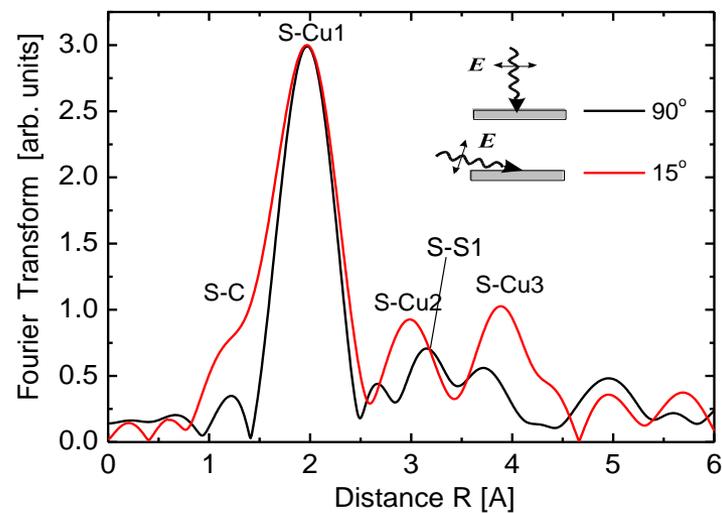
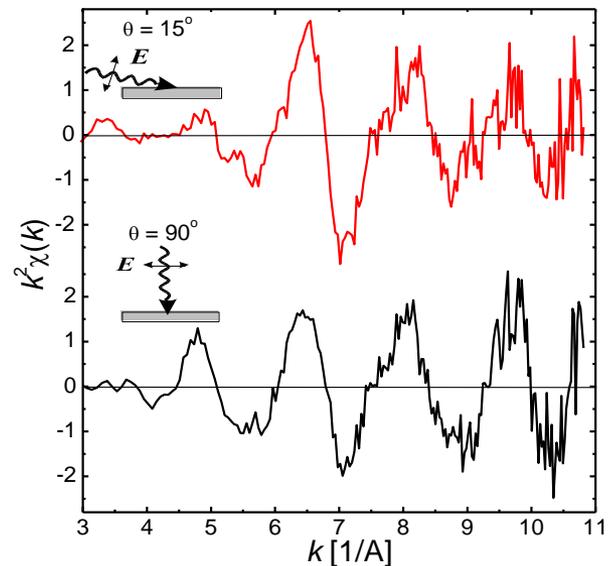


# Determination of Atomic Structure

## S K-edge EXAFS



## Surface-EXAFS



# Magnetic structures studied by XMCD

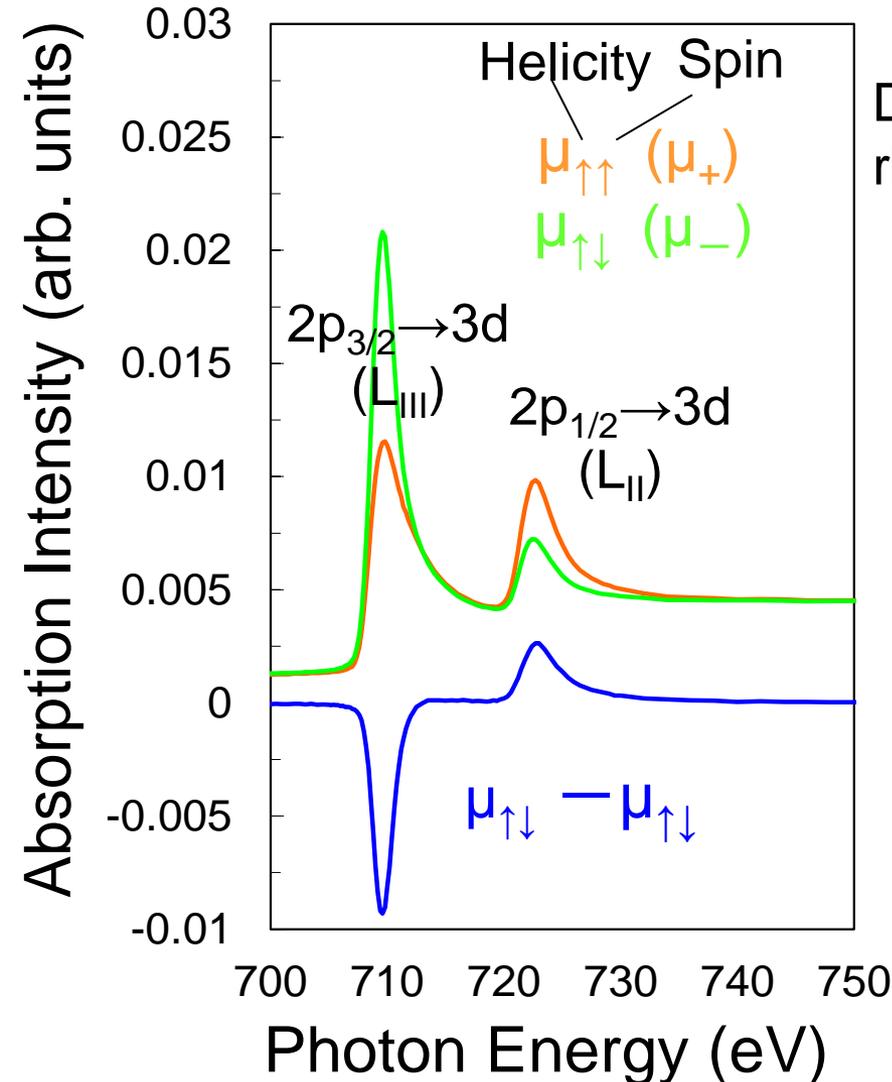
3 ML Fe / Cu(100)

Fe L-edge XMCD

X-ray Magnetic Circular Dichroism (XMCD)

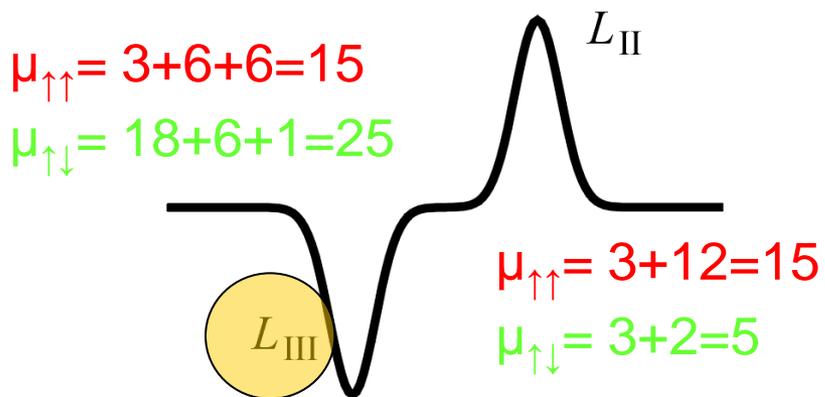
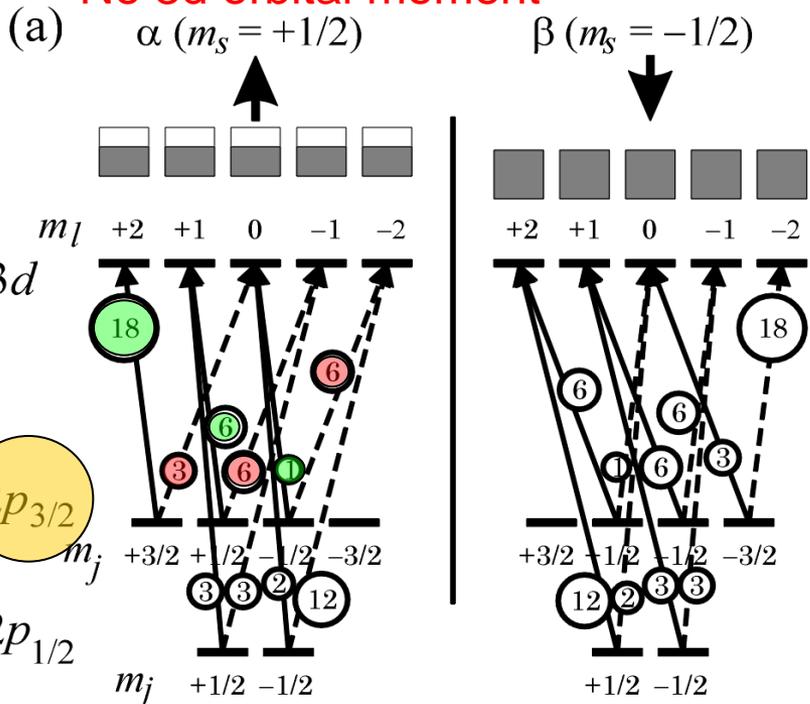
Difference in absorption intensities between right- and left-hand circular polarizations

1. **Element Selectivity**  
← resonant absorption ( $2p \rightarrow 3d \dots$ )
2. Determination of **Spin and Orbital magnetic moments**  
← **Sum rules**
3. High Sensitivity

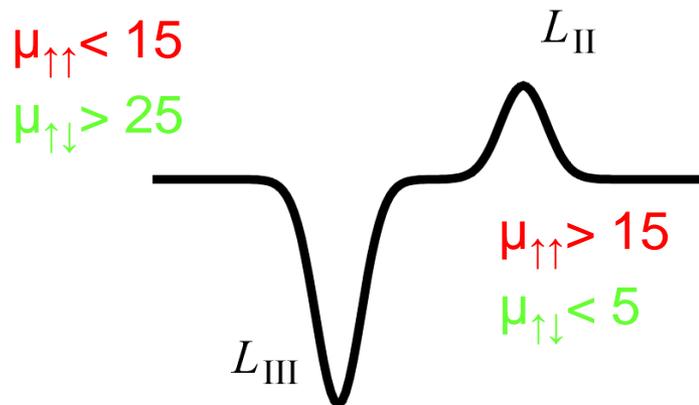
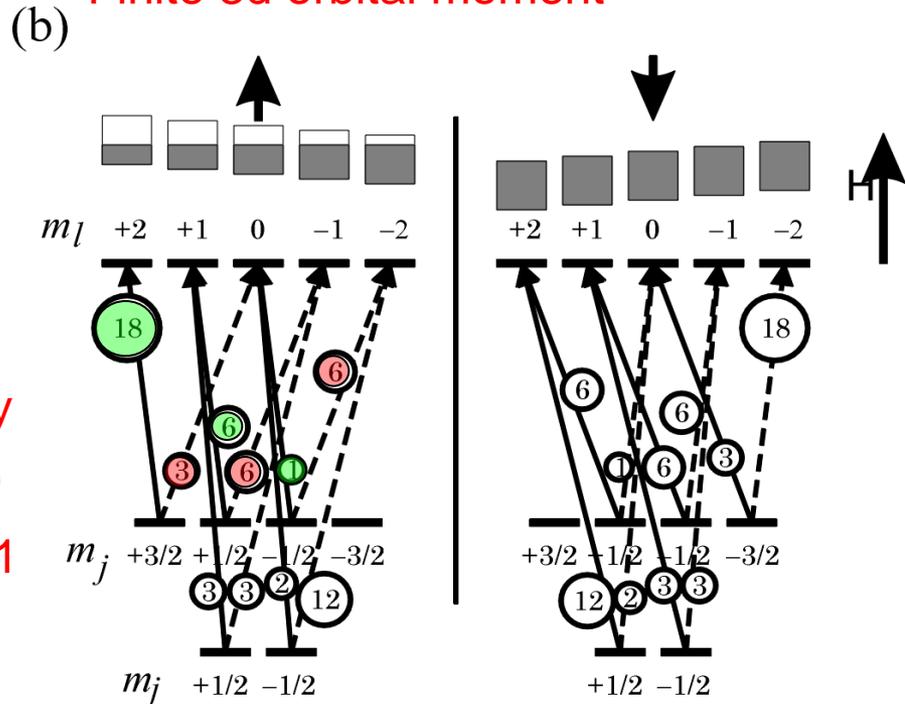


# Principle of XMCD

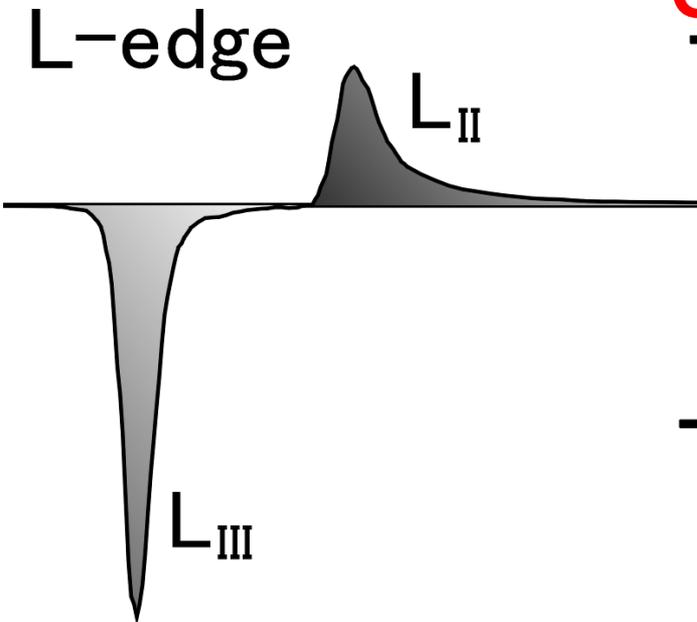
No 3d orbital moment



Finite 3d orbital moment



# XMCD Sum Rules



## Orbital moment $m_l$

$$\int L_{III}(\mu_+ - \mu_-) + \int L_{II}(\mu_+ - \mu_-)$$

$$\langle 0 \rightarrow m_l > 0$$

## Spin moment $m_s$

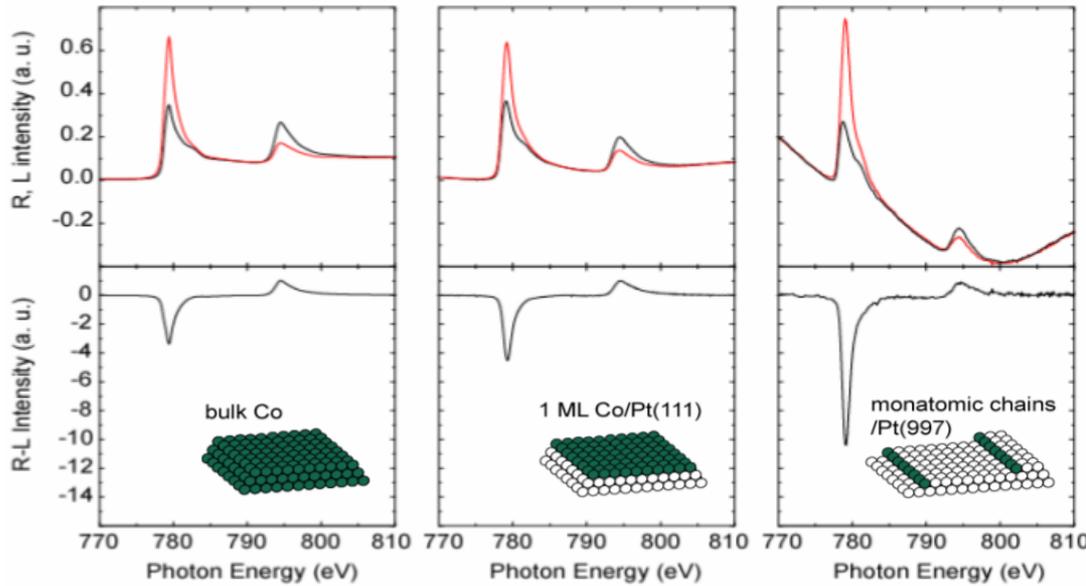
$$\int L_{III}(\mu_+ - \mu_-) - 2 \int L_{II}(\mu_+ - \mu_-)$$

$$\langle 0 \rightarrow m_s > 0$$

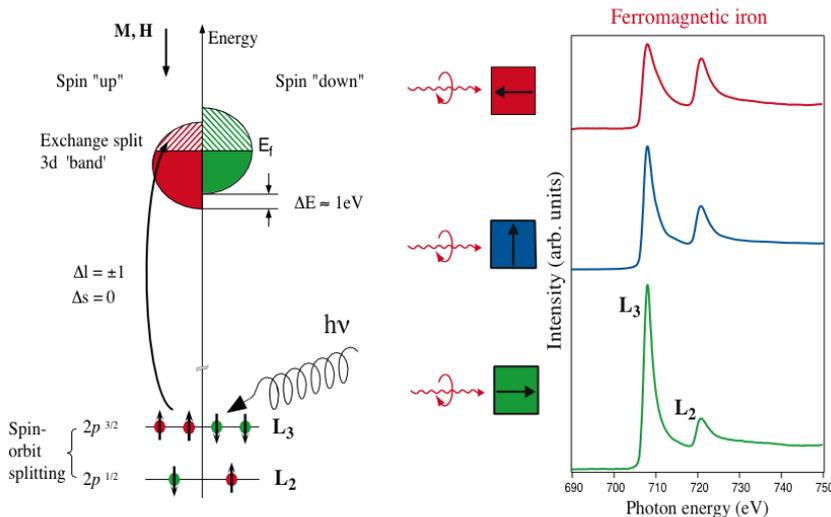
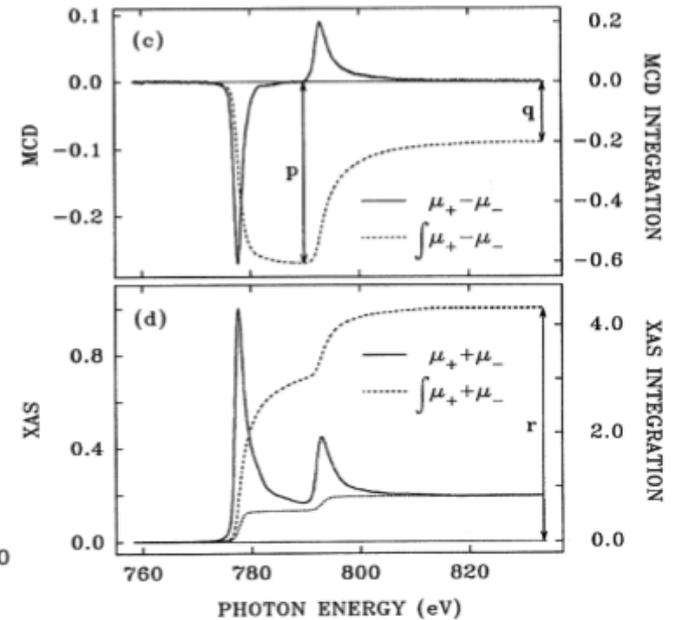
B.T. Thole et al., PRL **68**, 1943 (1992).  
P. Carra et al., PRL **70**, 694 (1993).

# Magnetism of Thin Films Studied by XMCD

## Co L-edge XMCD spectra



P. Gambardella, Nature **416**, 301 (2002)



## XMCD sum rules

$$m_{orb} = \frac{4q}{3r} (10 - n_{3d})$$

$$m_s = \frac{6p - 4q}{r} (10 - n_{3d})$$

$$\frac{m_{orb}}{m_s} = \frac{2q}{9p - 6q}$$

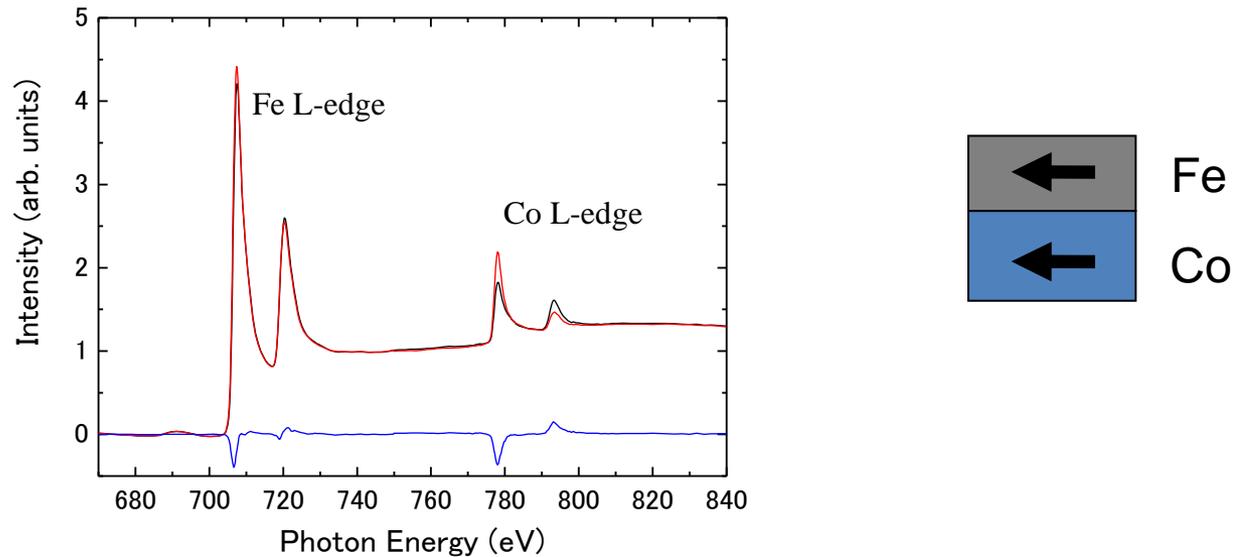
$$r = \int_{L_{2,3}} (\mu_+ + \mu_-) d\omega$$

$$p = \int_{L_3} (\mu_+ - \mu_-) d\omega$$

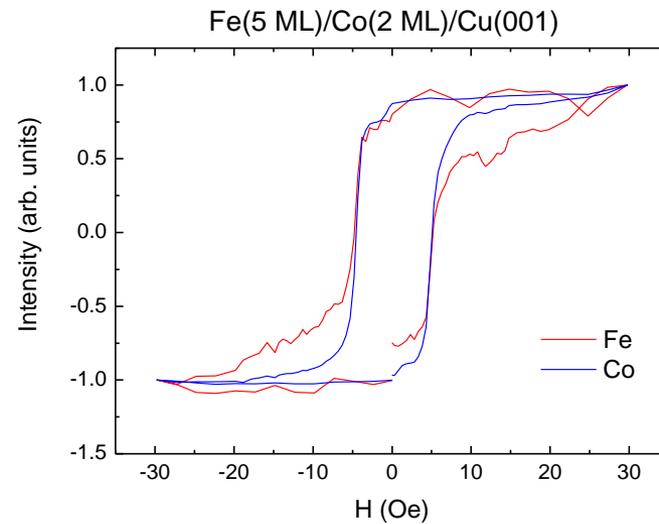
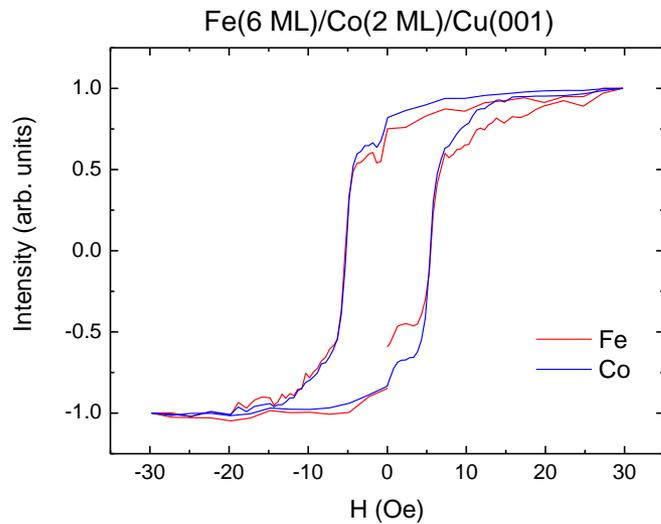
$$q = \int_{L_{2,3}} (\mu_+ - \mu_-) d\omega$$

C.T. Chen et al., PRL **75**, 152 (1995)

# Utilization of Element Selectivity of XMCD

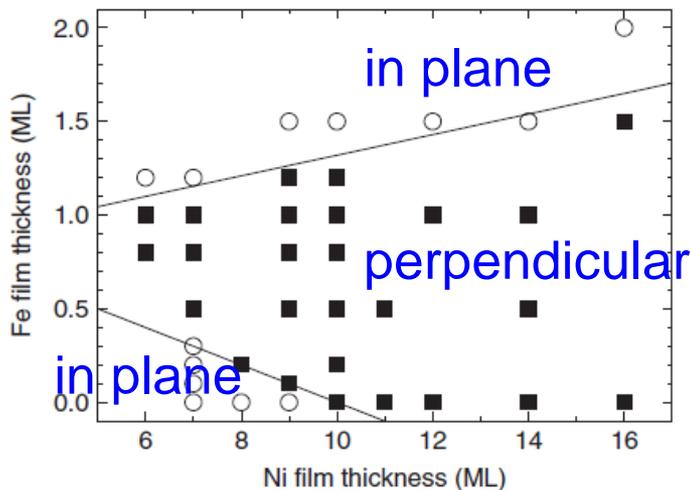
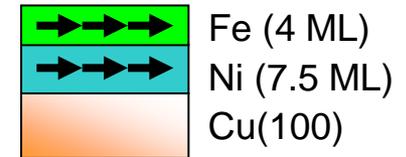
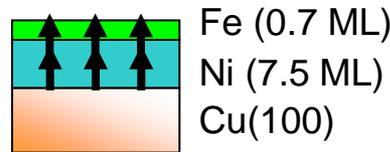
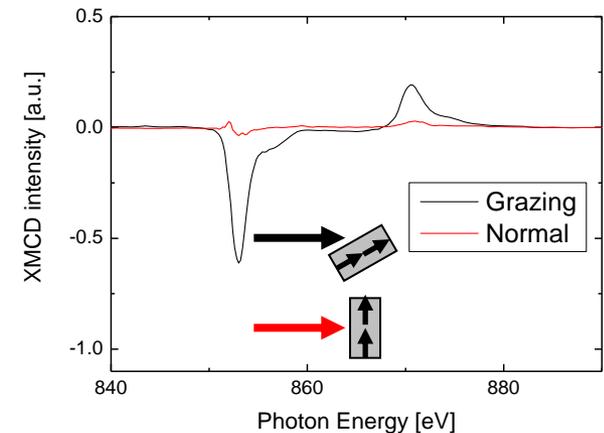
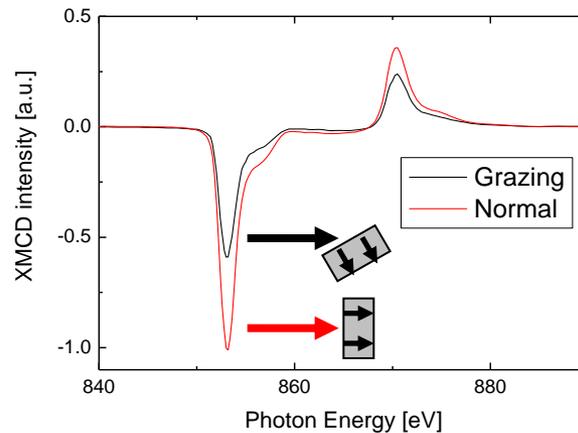
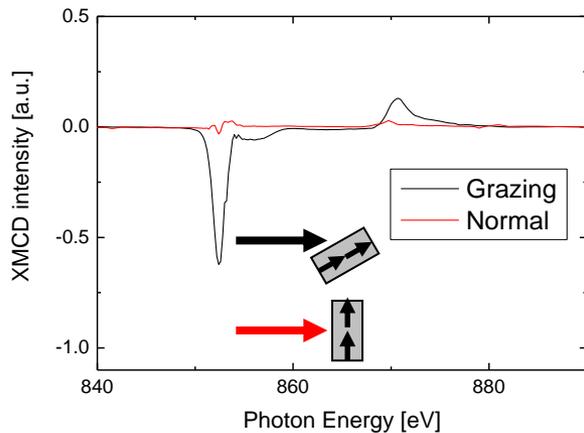


## Magnetic-field dependence of XMCD at Fe and Co L edges



# Angle Dependence of XMCD

## (1) weak magnetic field or remanent measurements



XMCD reflects magnetic component which is **parallel to X-ray beam**.

-> determination of easy axis of magnetization

Information on orbital moment

-> estimation of **magnetic anisotropy**

Abe et al., J. Magn. Magn. Mater. 206 (2006) 86.

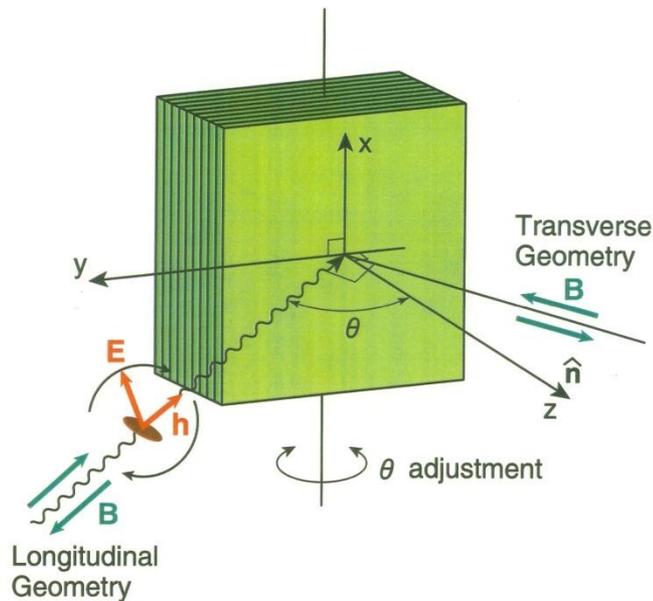
# Angle Dependence of XMCD

## (2) High magnetic field measurements

### XMCD (X-ray Magnetic Circular Dichroism)

Element selectivity

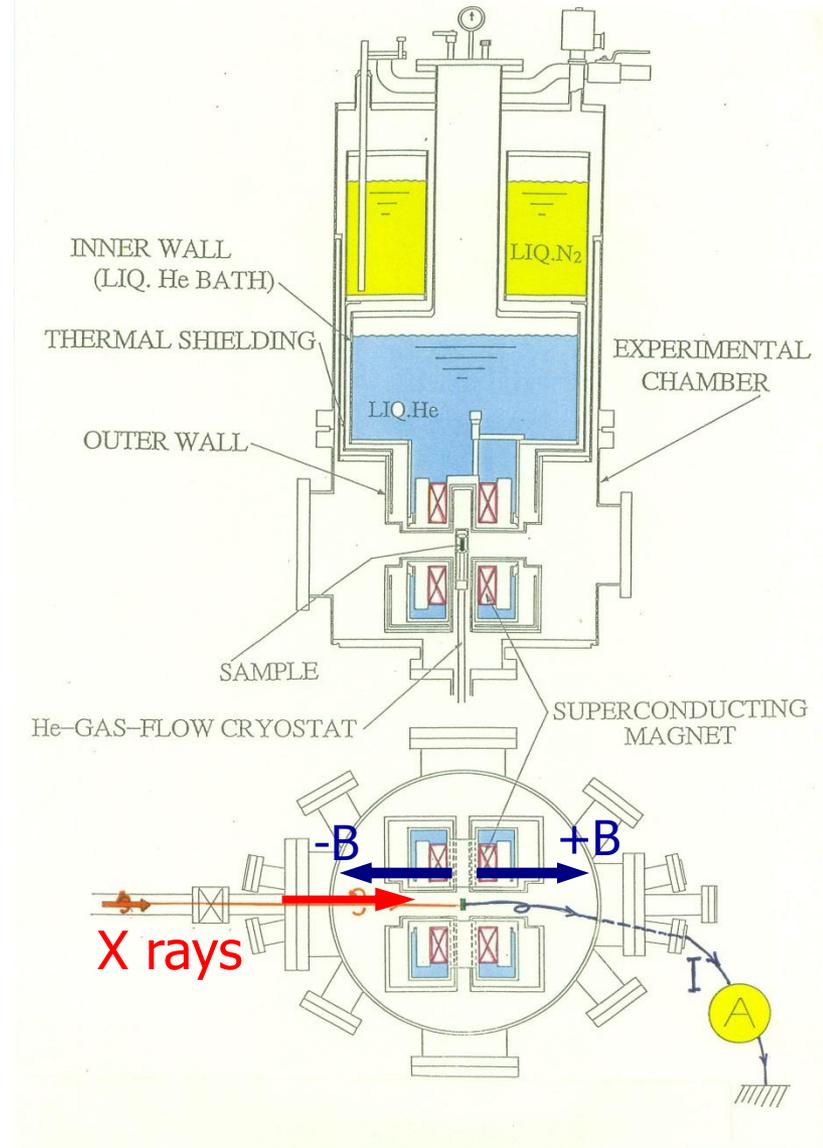
Quantitative determination of spin & orbital magnetic moments by using the sum rules



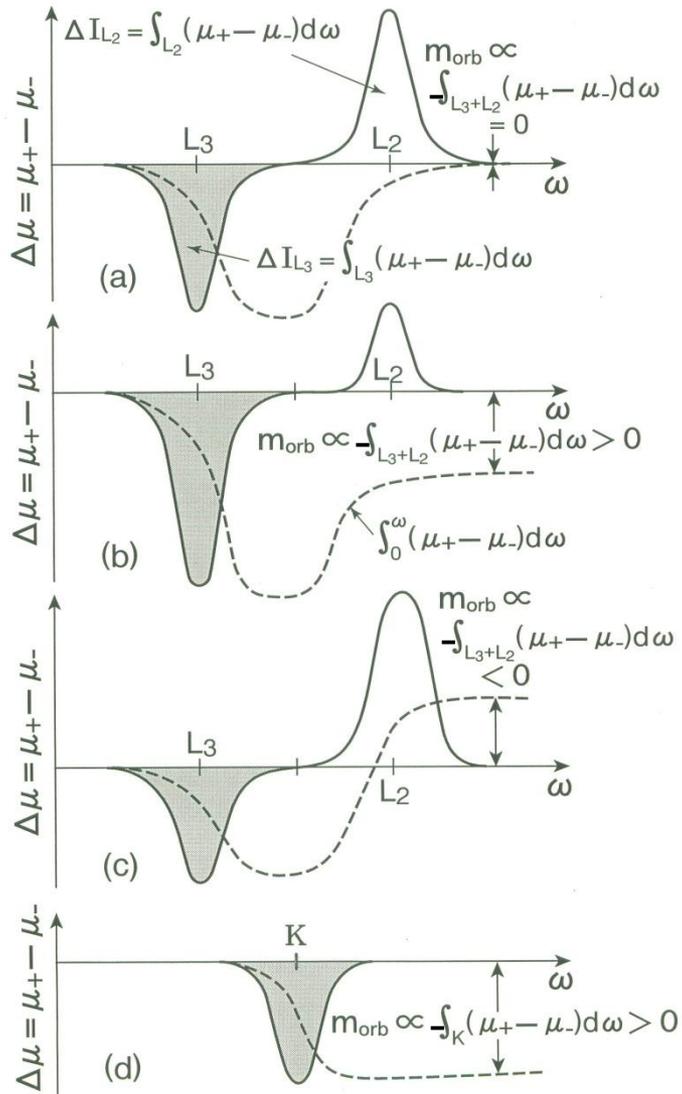
Angle-dependent XMCD

⇒ **Magnetic anisotropy**

**Separation of  $m_s$  from  $m_T$**



# Angle-dependent XMCD Sum Rules



## Orbital sum rule

$$\frac{[\Delta I_{L_3} + \Delta I_{L_2}]^\theta}{I_{L_3} + I_{L_2}} = -\frac{3 \cdot m_l^\theta}{4n_h \cdot \mu_B}$$

## Spin sum rule

$$\frac{[\Delta I_{L_3} - 2 \cdot \Delta I_{L_2}]^\theta}{I_{L_3} + I_{L_2}} = -\frac{(m_s + 7 \cdot m_T^\theta)}{2n_h \cdot \mu_B}$$

B.T. Thole et al., PRL **68**, 1943 (1992).  
 P. Carra et al., PRL **70**, 694 (1993).

$$m_l^\theta = m_l^\perp \cos^2 \theta + m_l^\parallel \sin^2 \theta$$

$$m_T^\theta = m_T^\perp \cos^2 \theta + m_T^\parallel \sin^2 \theta$$

# Investigation of Interface Magnetism

## Au/Co(2 ML)/Au(111)

Self-assembled Co islands  
due to a reconstruction of  
Au surface

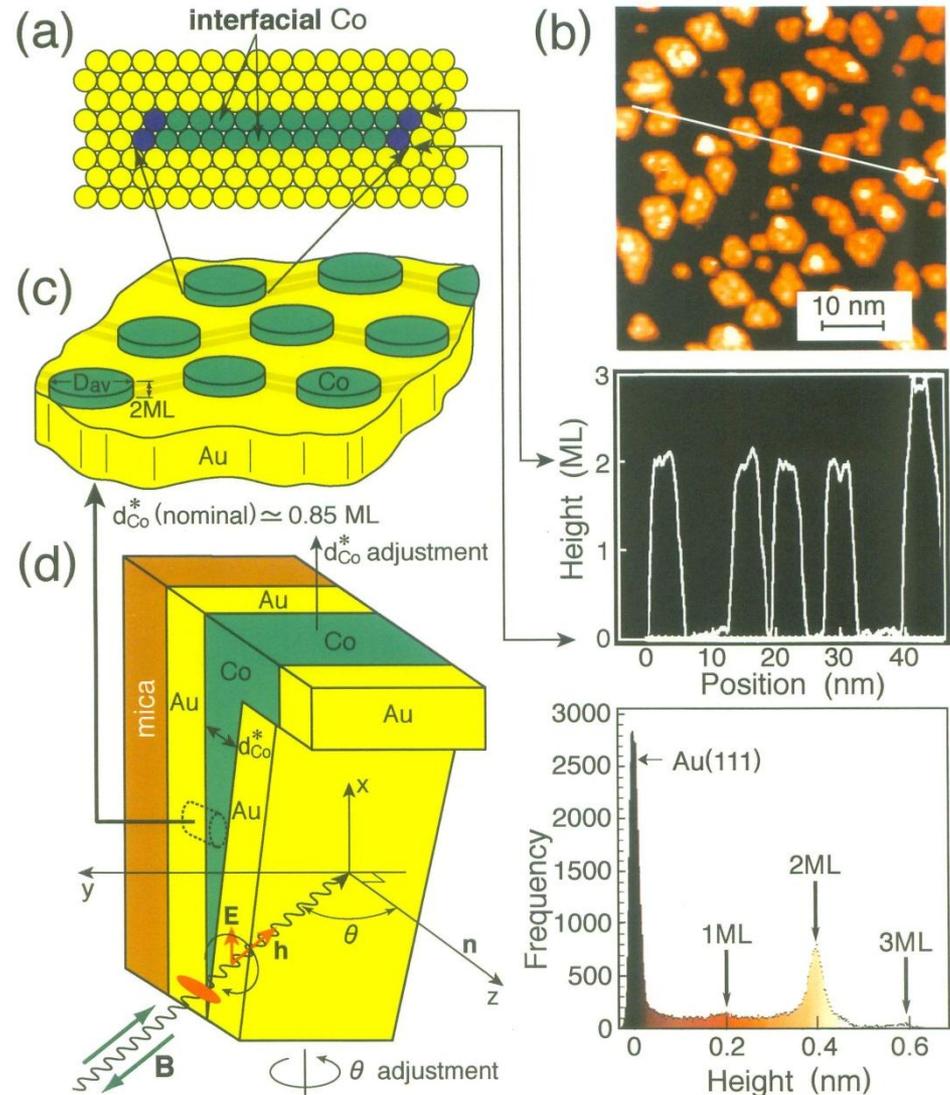
All Co atoms are regarded to  
“interface” because of 2 ML  
thickness

⇒ **Direct observation of  
interface magnetism**

Angle-dependent XMCD

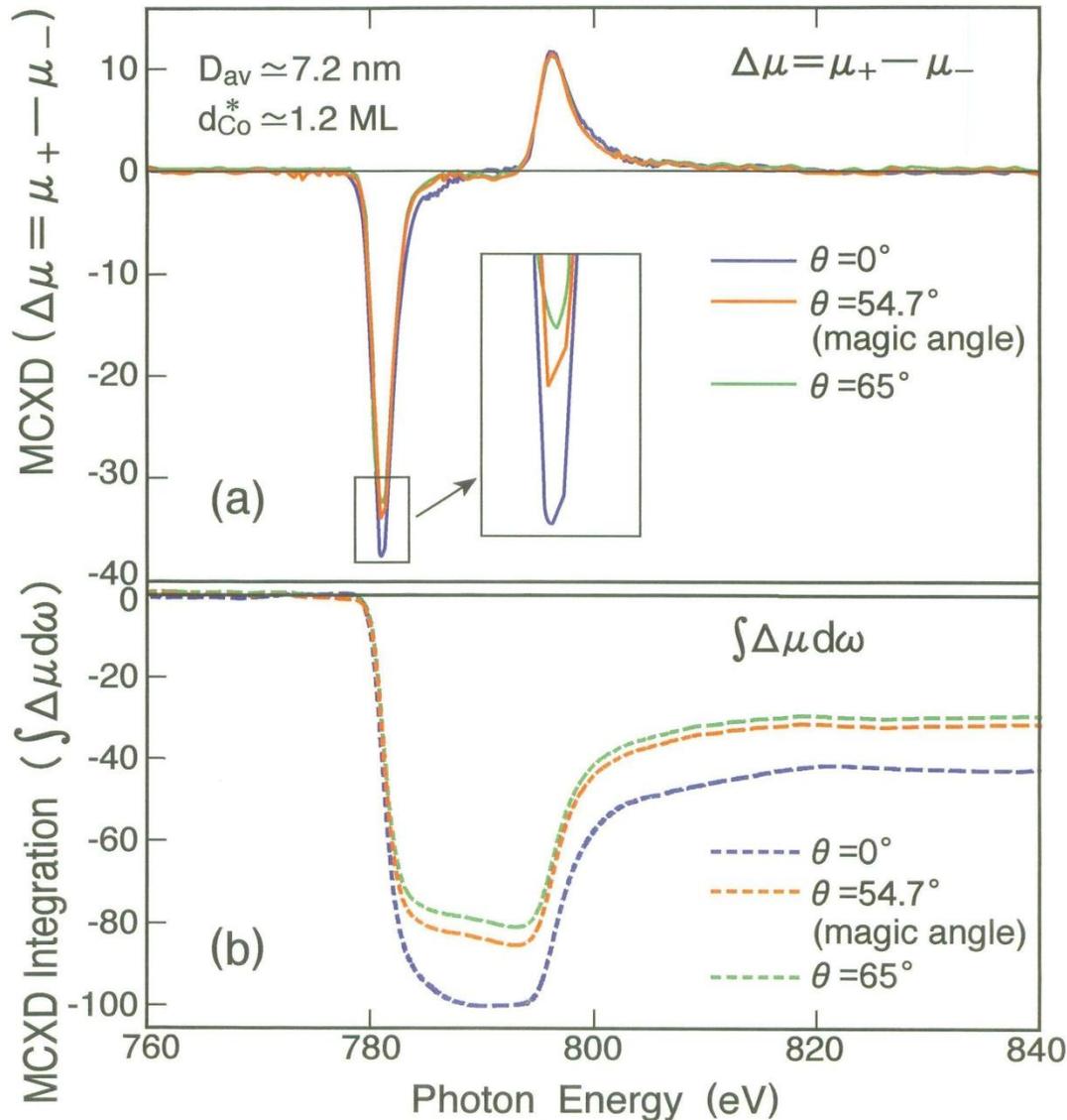
⇒ **Direct determination of  
 $m_s$ ,  $m_l^{\parallel}$ ,  $m_l^{\perp}$ ,  $m_T^{\parallel}$ ,  $m_T^{\perp}$**

T. Koide et al., Phys. Rev. Lett. 87, 257201 (2001)



# Angle-dependent XMCD Measurements

T. Koide et al., Phys. Rev. Lett. 87, 257201 (2001)



## PF BL-11A

### Angle dependence in XMCD

← Anisotropy in  $m_l$ ,  $m_T$

$$m_j^\theta = m_j^\perp \cos^2 \theta + m_j^\parallel \sin^2 \theta$$

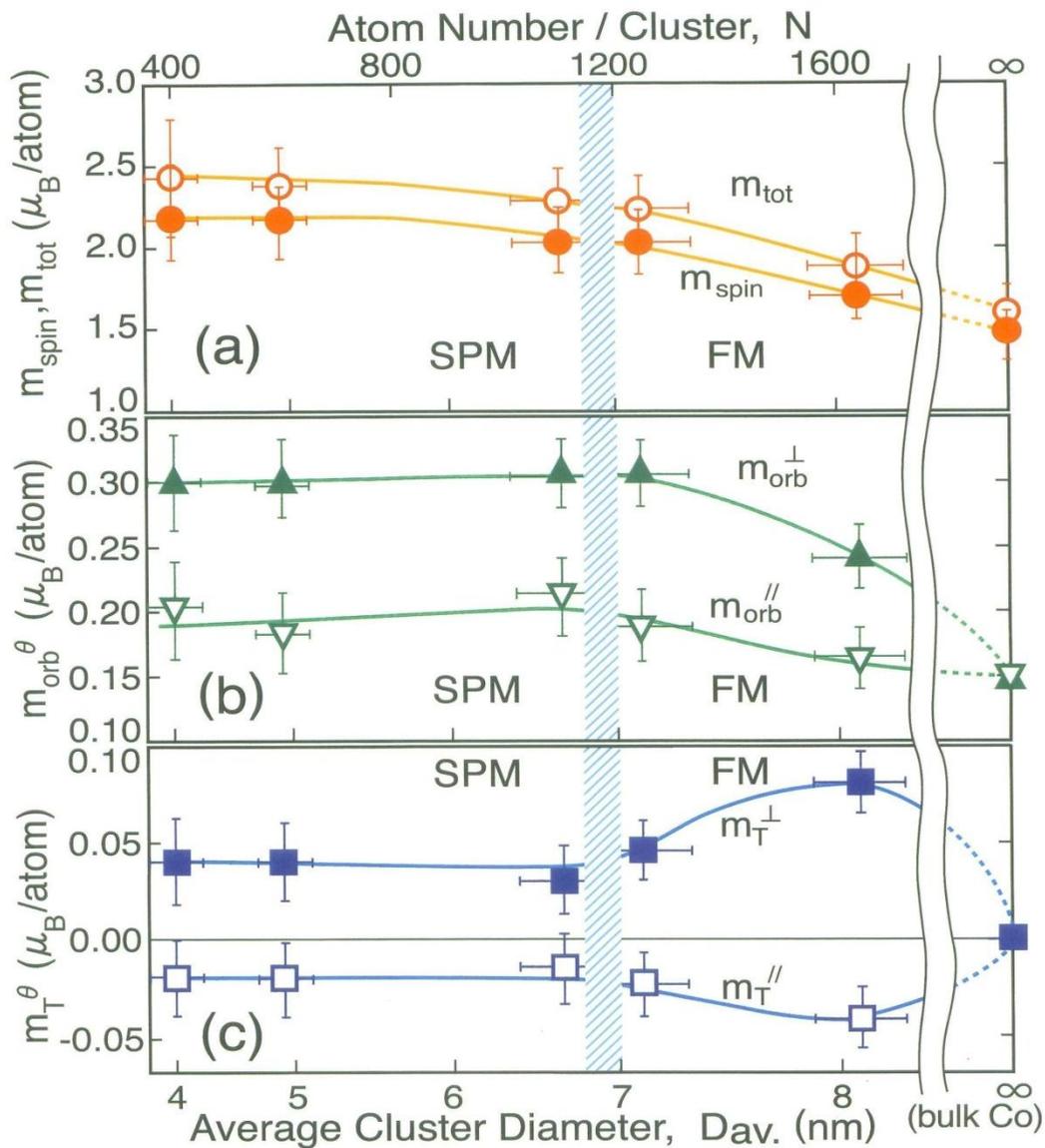
(j = l or T)

$$m_T^\perp + 2 m_T^\parallel = 0$$

⇒ Determination of all moments including their **anisotropy**

# Determined Magnetic Moments

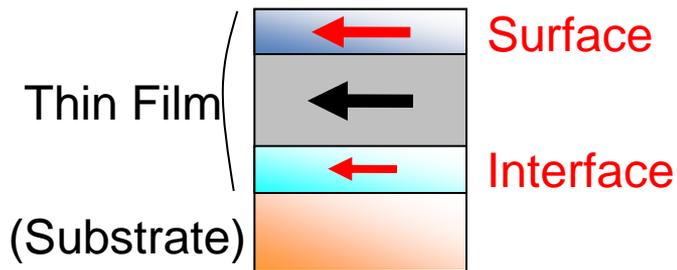
T. Koide et al., Phys. Rev. Lett. 87, 257201 (2001)



**Cluster-size dependent phase transition**

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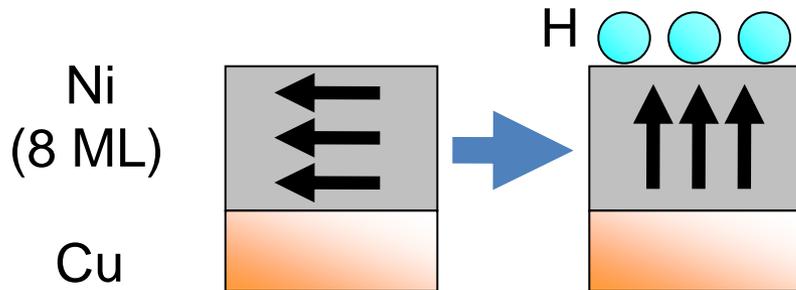
# Introduction: Exploring Magnetic Depth Profile



Surface and interface show different magnetism from inner layers

Surface and Interface sometimes affect magnetism of whole film

Surface effect: Gas adsorption



Vollmer, et al., Phys. Rev. B 60 (1999) 6277.

Adsorption-induced change in magnetic easy axis  
⇒ What happens at surface?

Interface effect: TMR

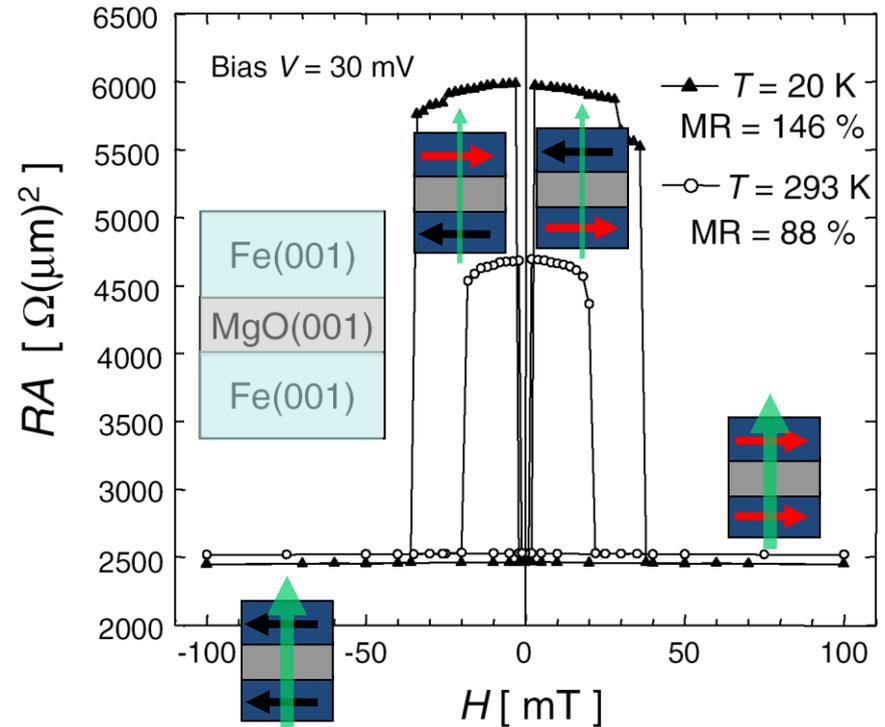


Fig. 3. Magnetoresistance curves for Fe(001)/MgO(001)(20 Å)/Fe(001) MTJ at  $T = 293$  and 20 K. The MR ratios were 88% and 146%, respectively.

Yuasa, et al., Jpn. J. Appl. Phys. 43 (2004) L588.

Chemical and magnetic states at interface affect MR ratio

# Co/Cu(100) - Surface & interface orbital moment -

Tischer et al., PRL 75 (1995) 1602.

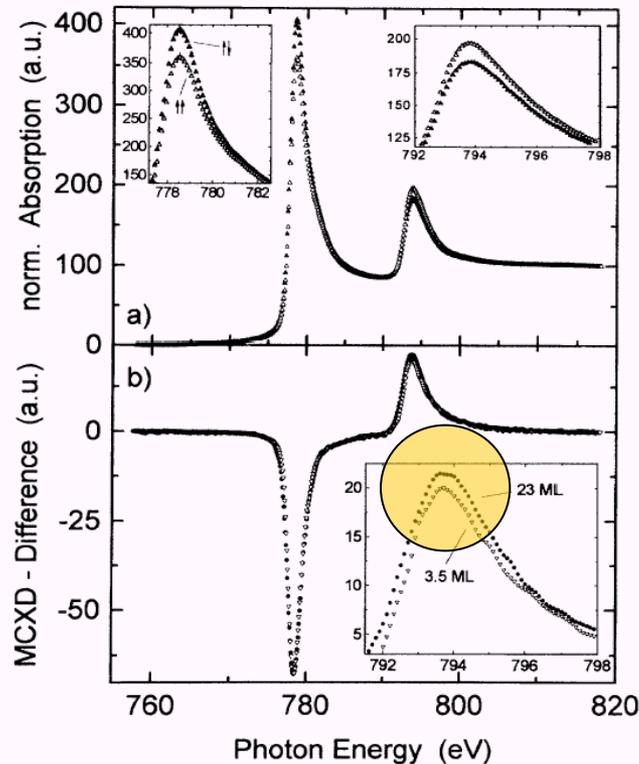


FIG. 2. (a) The normalized absorption spectrum for 3.5 ML Co/Cu(100). Open triangles indicate the photon spin parallel to the remanent magnetization, full triangles antiparallel. (b) MCXD difference for the 3.5 ML film (triangles) and a thick 23 ML film (circles). Both are normalized to the same  $L_3$  intensity to demonstrate that the dichroic response around the  $L_2$  edge is relatively smaller for the thin film.

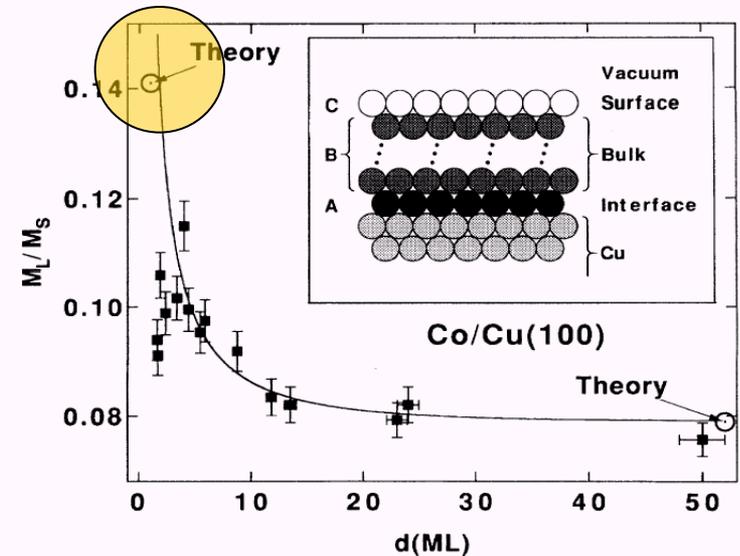


FIG. 3. The ratio of orbital versus spin moment  $M_L/M_S$  as a function of film thickness  $d$ . The open circles give the theory taken from Table I. The full squares show the experiment. The solid line is a fit using Eq. (1) with the parameters given in the last row of Table I. Note that the fit was performed only for  $d \geq 3$  ML, corresponding to well-defined, epitaxial growth. The surface, interface, and bulk contributions used in Eq. (1) are schematically shown in the inset.

# [Pt/Co(111)] multilayer - Interface orbital moment -

Nakajima et al., PRL 81 (1998) 5231.

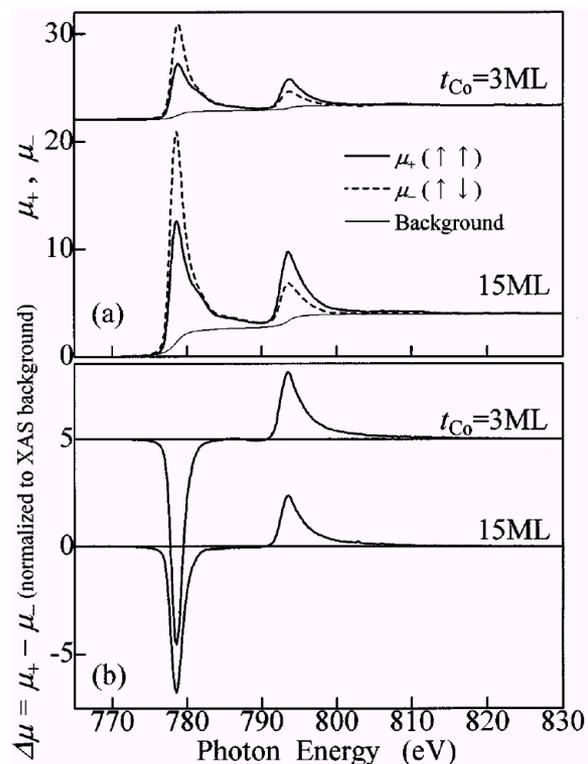


FIG. 2. (a) Polarization-dependent Co  $L_{2,3}$  XAS spectra of  $\text{Co}(t_{\text{Co}})/\text{Pt}(7.5 \text{ ML})$  multilayers for  $t_{\text{Co}} = 3$  and 15 ML. Correction for  $P_C$  was made. The thin solid curve denotes the averaged XAS background. (b) Co  $L_{2,3}$  MCXD spectra normalized by the edge jump above 820 eV in XAS of (a).

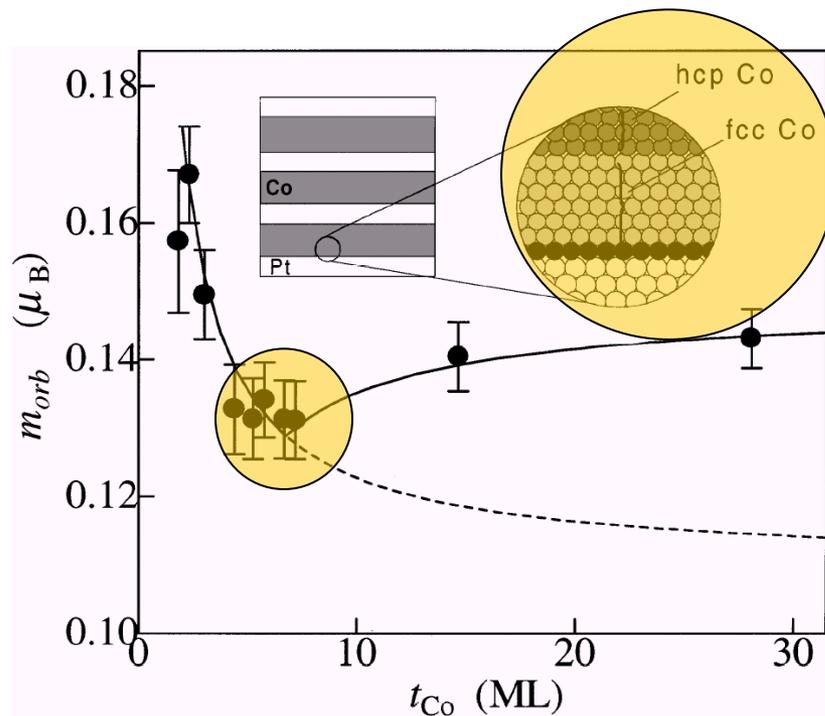


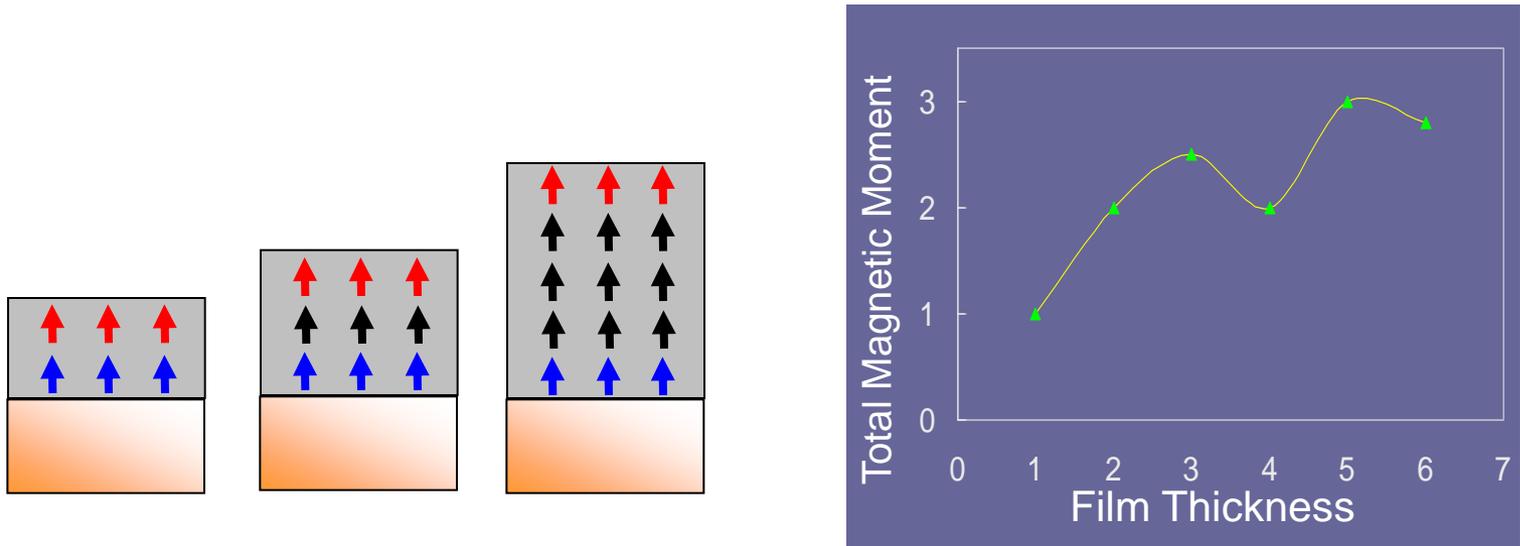
FIG. 3. Orbital magnetic moment of Co perpendicular to the film plane as determined from the MCXD and XAS spectra of Fig. 2 using the MCXD orbital sum rule. The solid curve represents a fit to the data based on the model shown in the inset. The dashed curve denotes an extrapolation for a supposed case of all fcc Co layers.

# Conventional Technique for Depth Profiling

SQUID, VSM, MOKE, XMCD...

Gives **averaged information** over the whole sample.

⇒ also **averaged in depth**



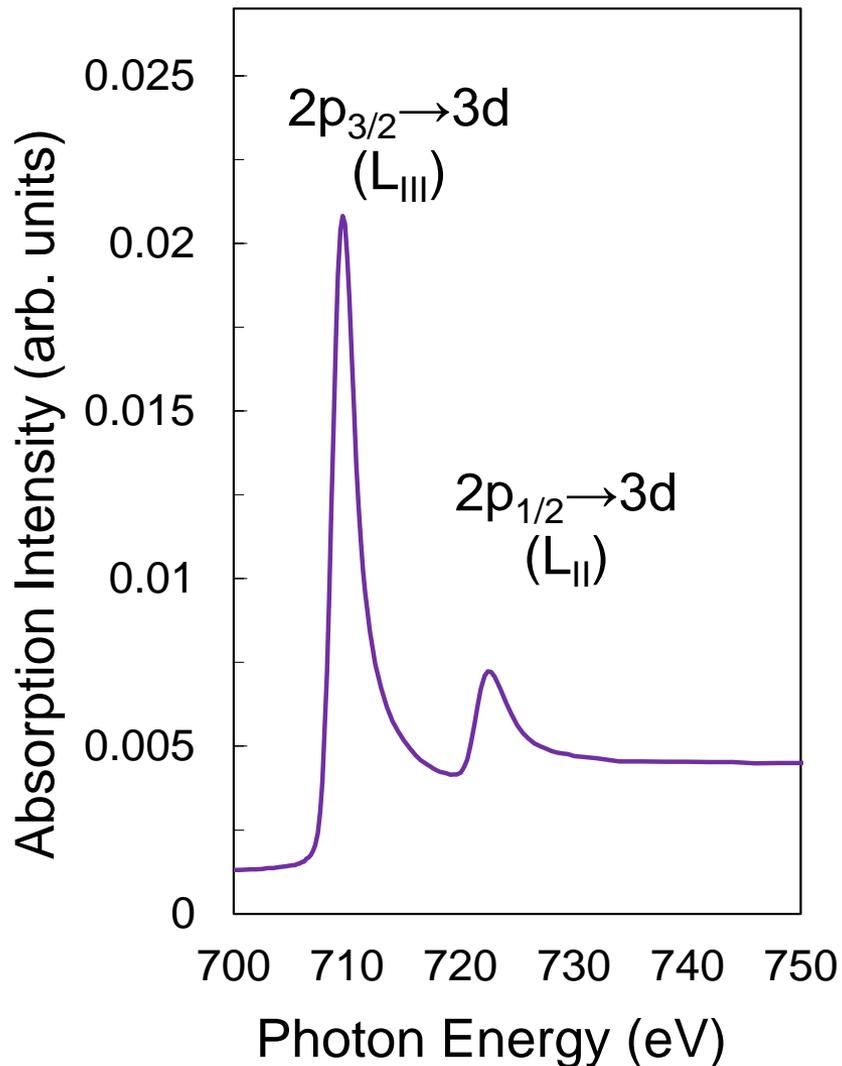
Based on an **assumption**  
that magnetic structure of surface and interface  
**dose not change** upon layer growth



**Direct technique for depth profiling**

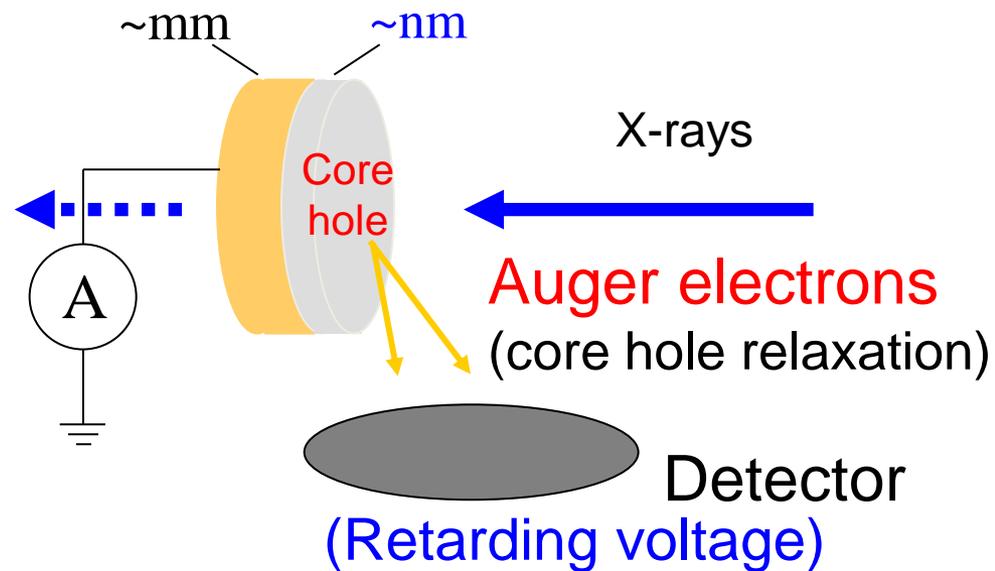
# XAS Measurement in the Soft X-ray Region

3 ML Fe / Cu(100) **Fe L-edge XAS**



How can we measure

**X-ray absorption** spectrum ?



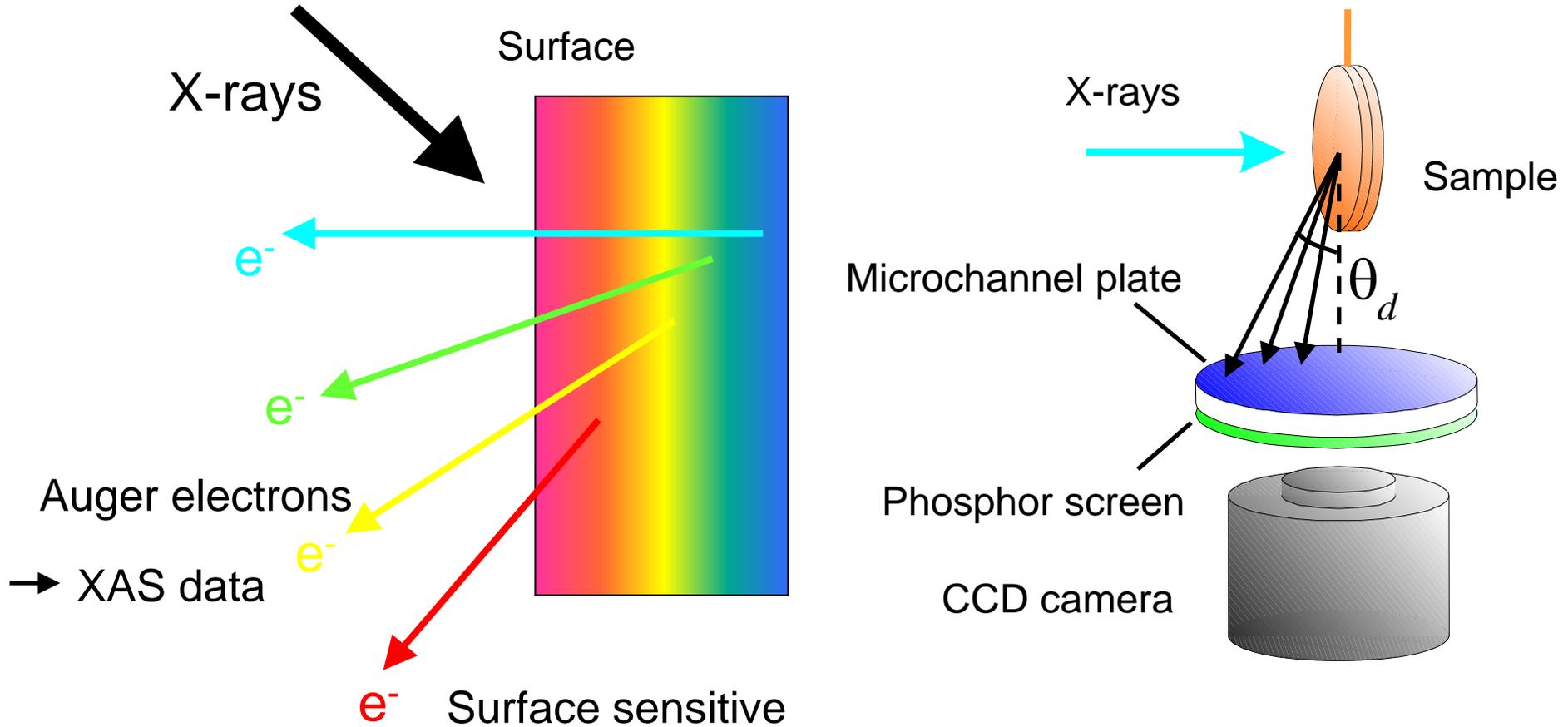
**Electron yield** XAS

Total electron yield (TEY)

Partial electron yield (PEY)

cf. **Fluorescence yield** (FY)

# Principle of Depth-resolved XAS



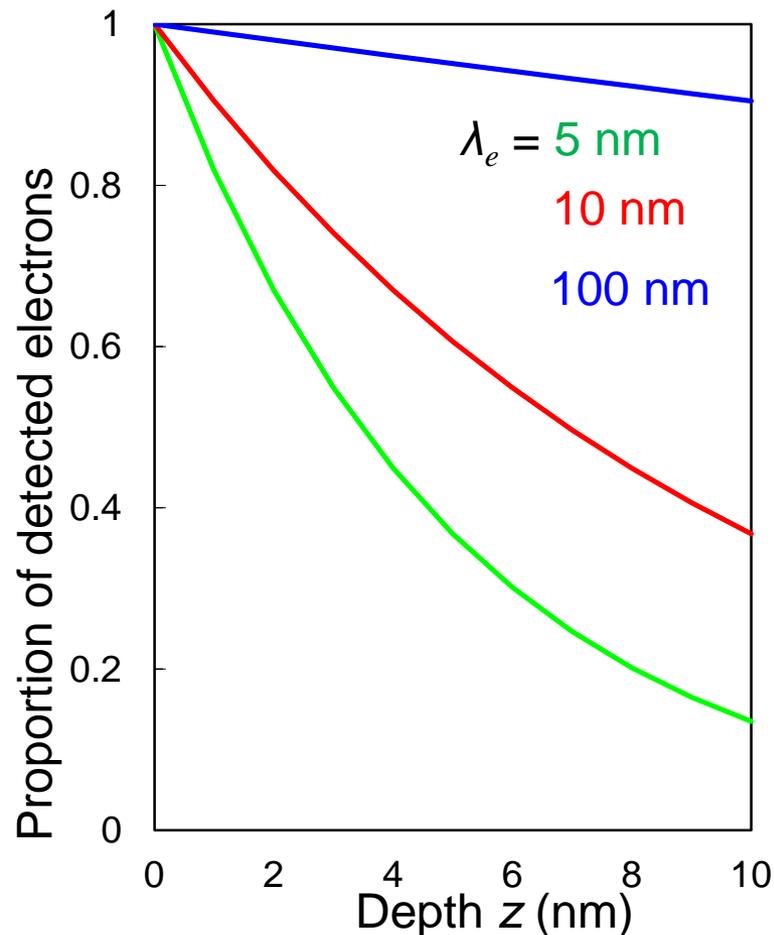
Electron yield XAS measurements at different detection angles

→ A set of XAS data with different probing depths

# Probing Depth (effective escape depth): $\lambda_e$

Number of detected electrons emitted at depth  $z$ :  $I = I_0 \exp(-z/\lambda_e)$

$I_0$ : Original number of emitted electrons



**Small  $\lambda_e$**

$\Rightarrow$  Large contribution from **surface**

XAS:

Averaged information **per atom**

**Depth-resolved XAS:**

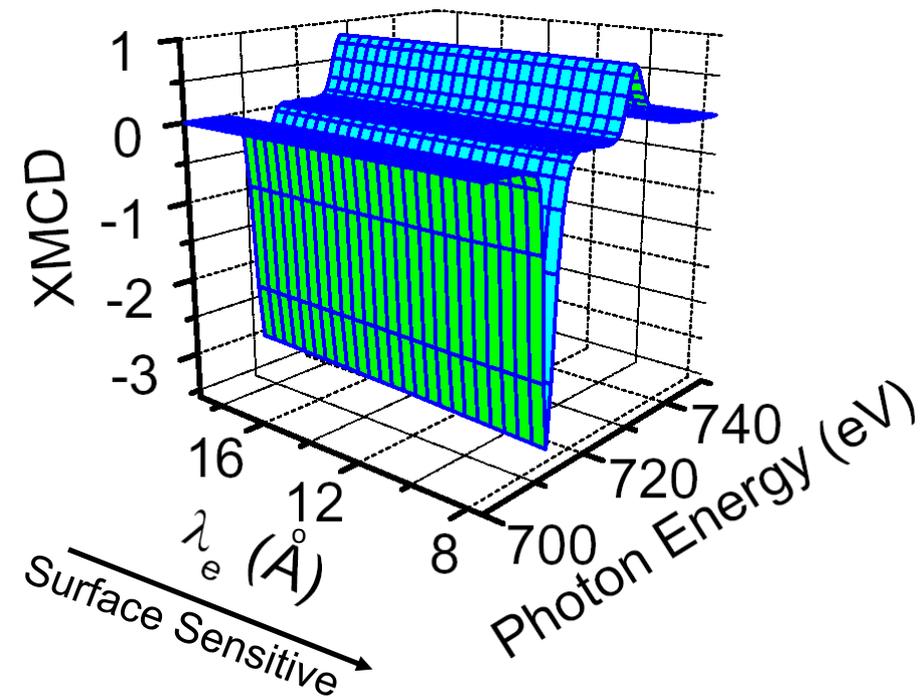
$\exp(-z/\lambda_e)$ -**weighted average**

# Feasibility Study: Depth-resolved XMCD of Fe/Cu(100)

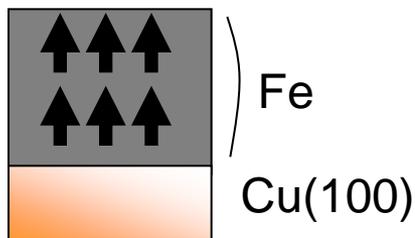
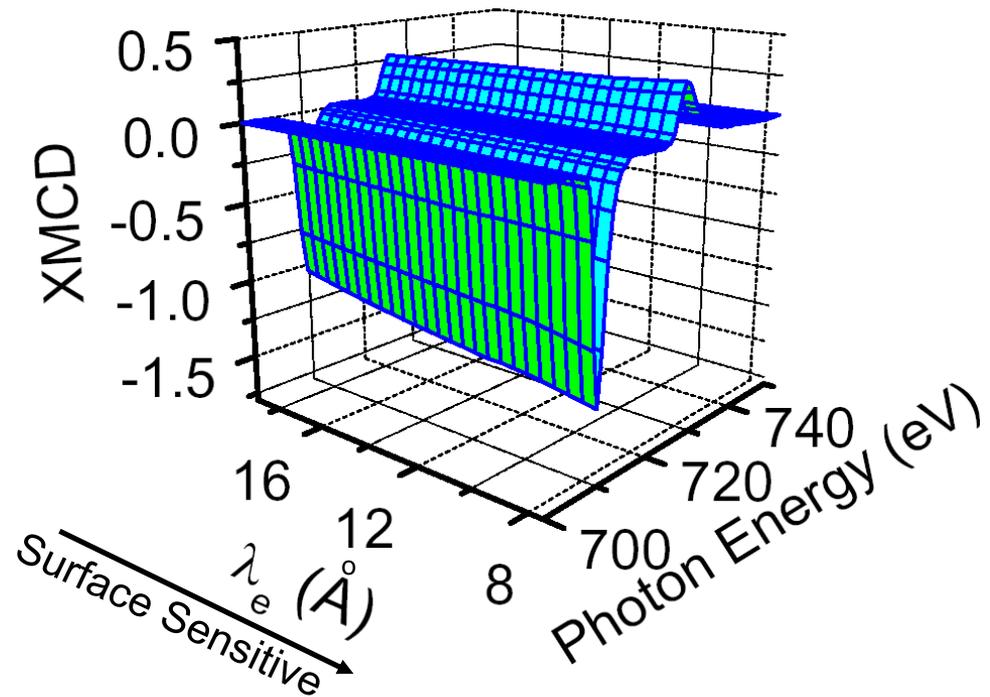
Amemiya et al., APL 84 (2004) 936.

Normal Incidence, 130 K

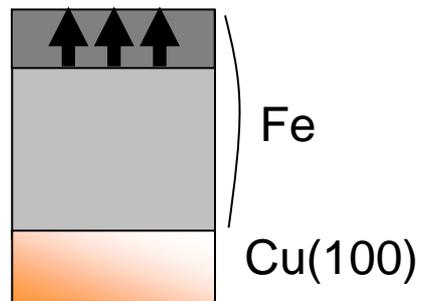
## 3 ML Fe



## 7 ML Fe

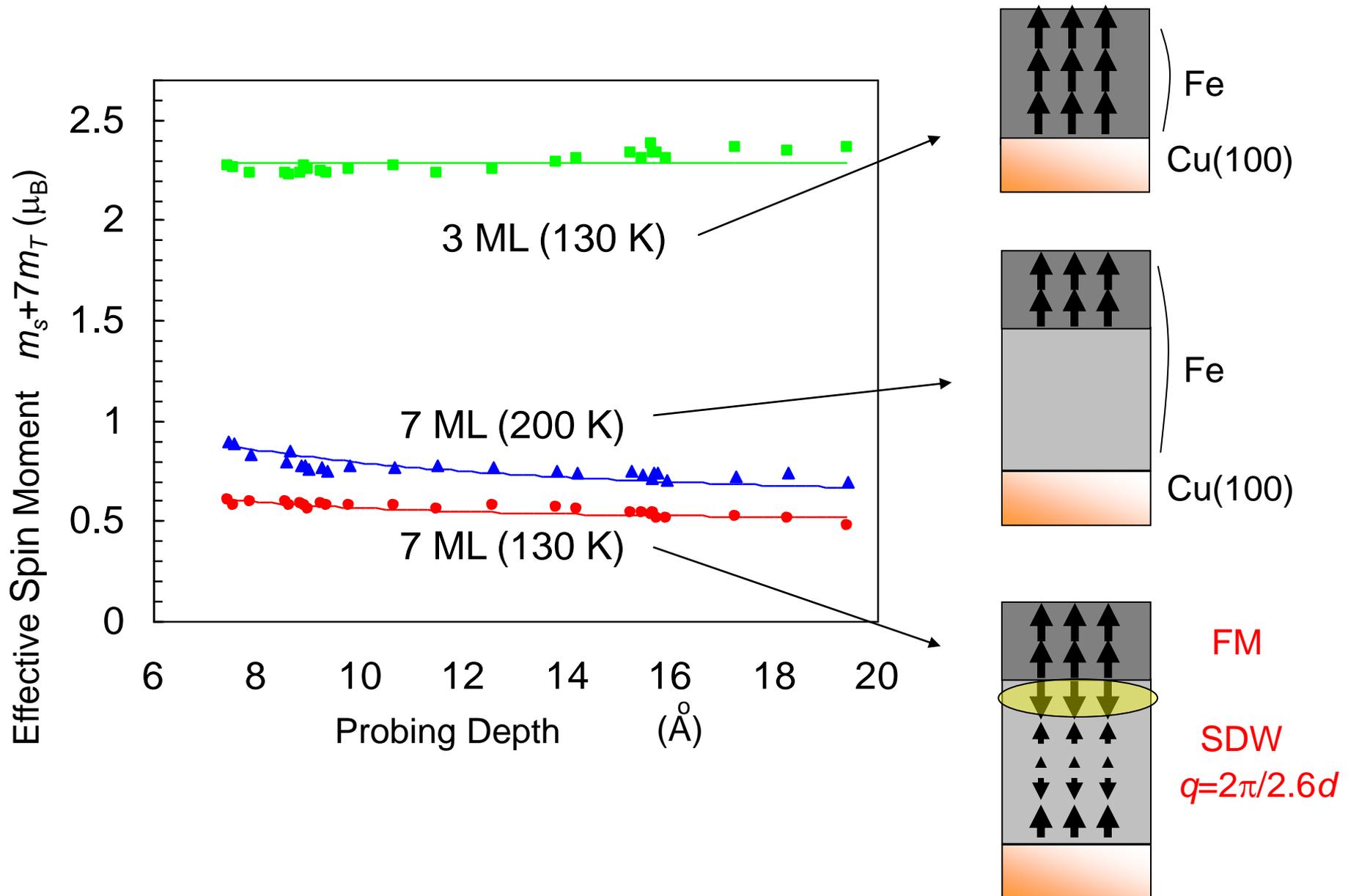


Uniform  
Magnetization

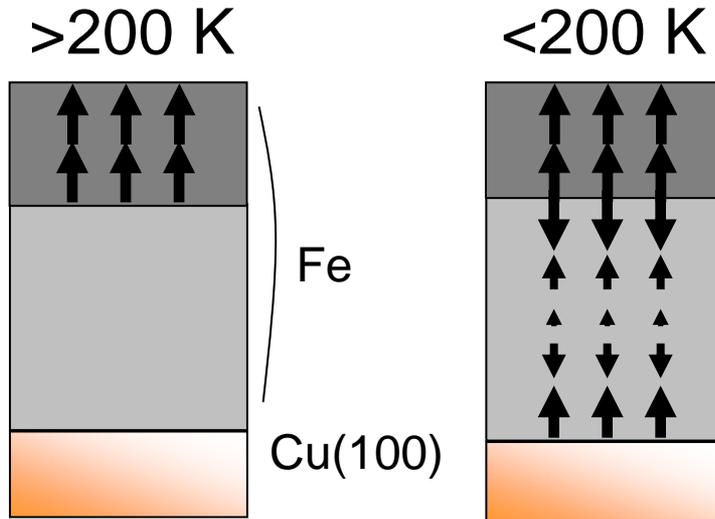


Surface  
Magnetization

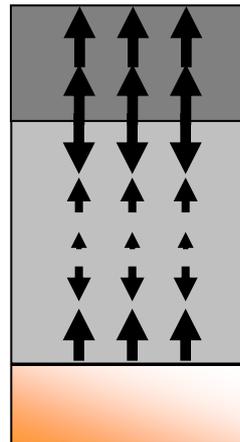
# Interpretation of depth-resolved XMCD data



## Fe/Cu(100)



<200 K



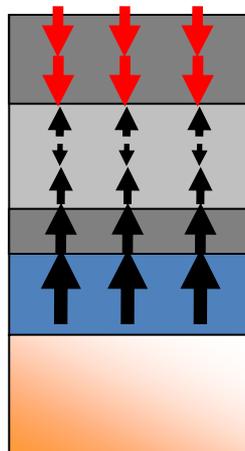
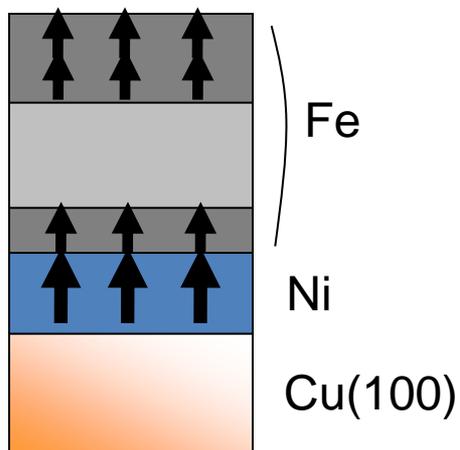
Surface (FM)

Inner layers (AFM or SDW)

No (little) magnetic interaction between

Cu and interface (bottom) Fe

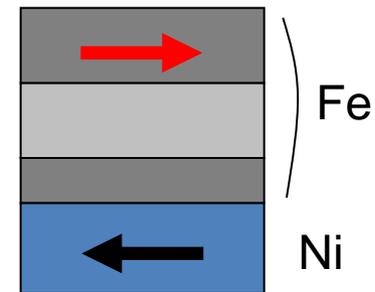
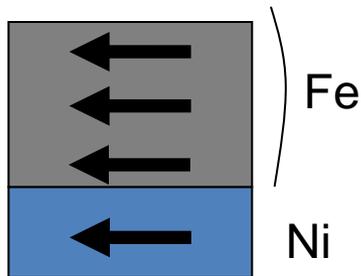
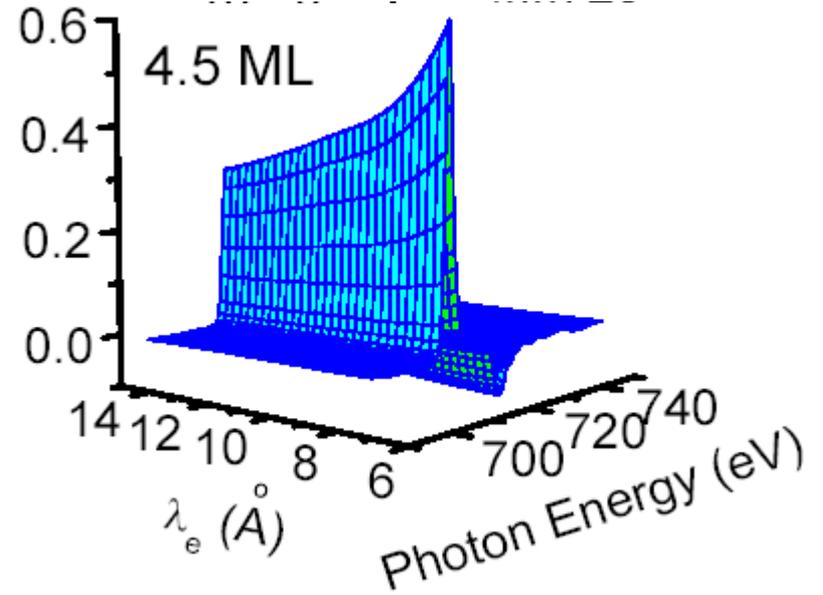
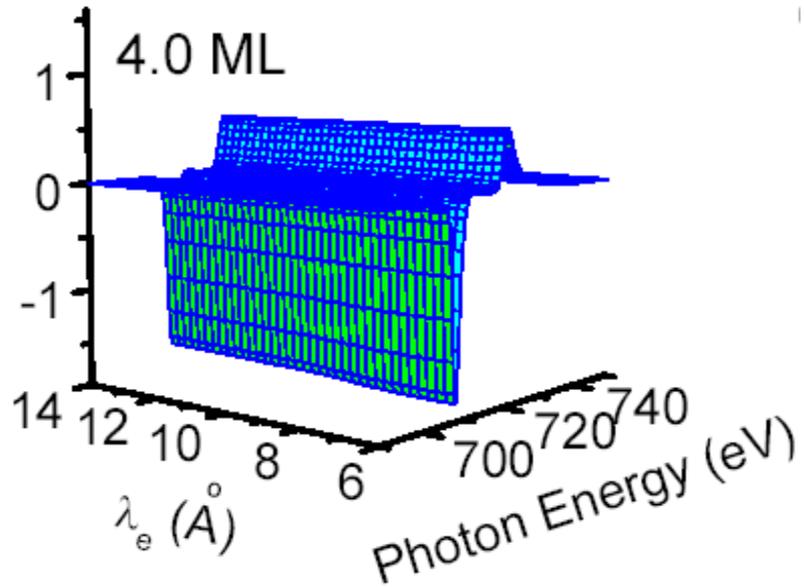
## Fe/Ni/Cu(100)

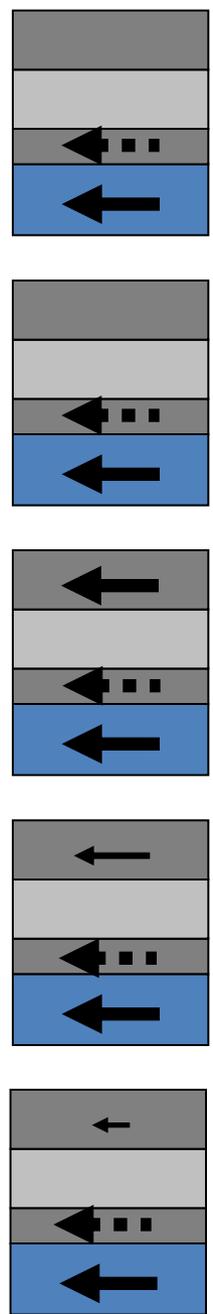
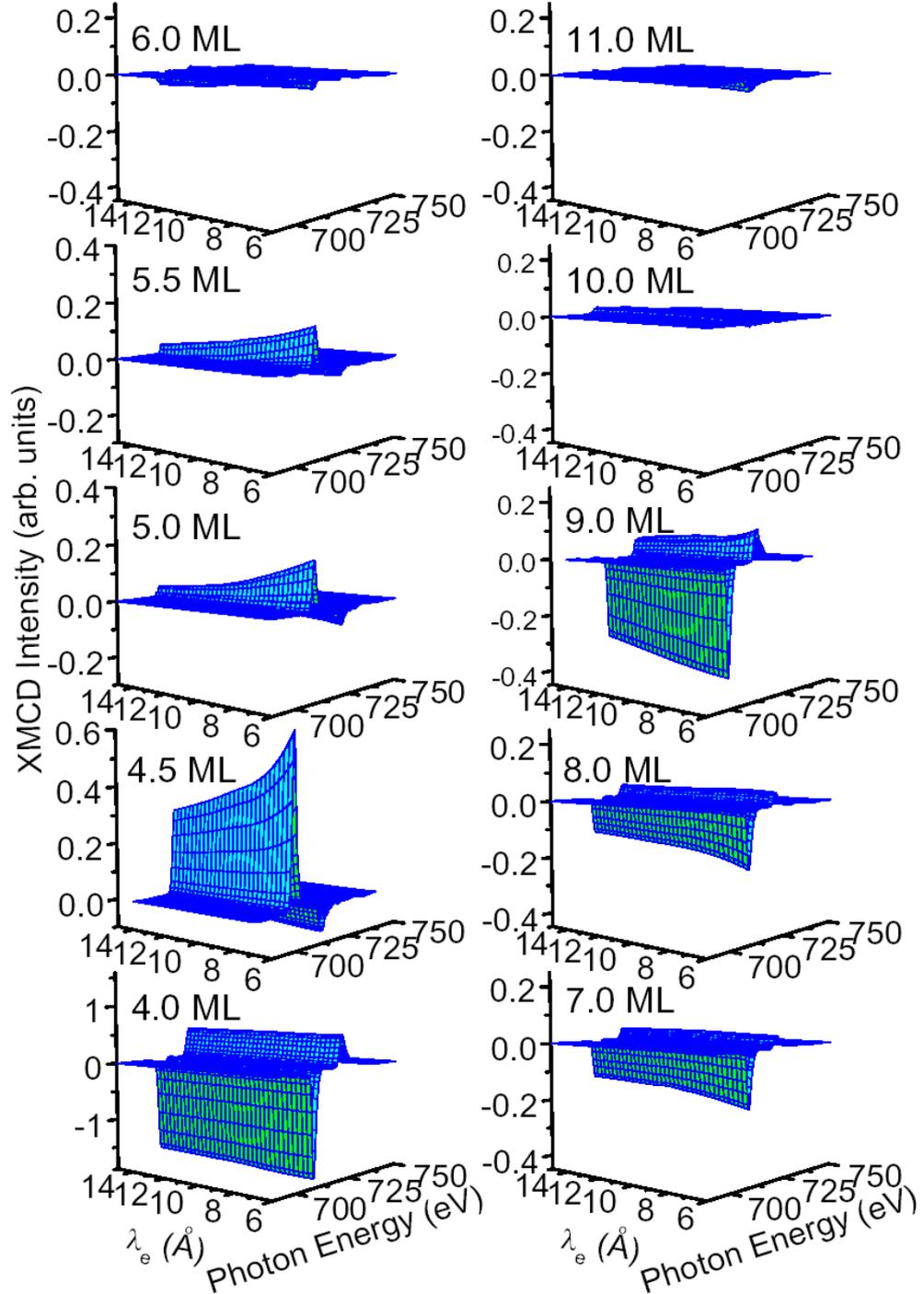
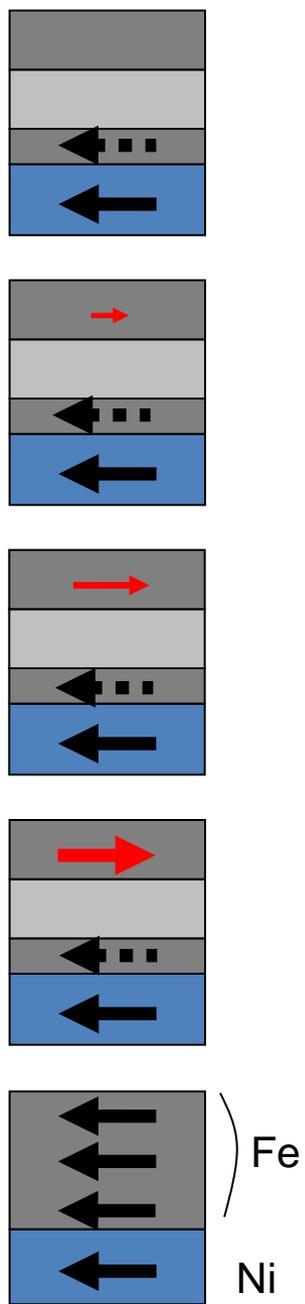


Any magnetic interaction  
among **surface**, **inner layers**  
and **interface** ?

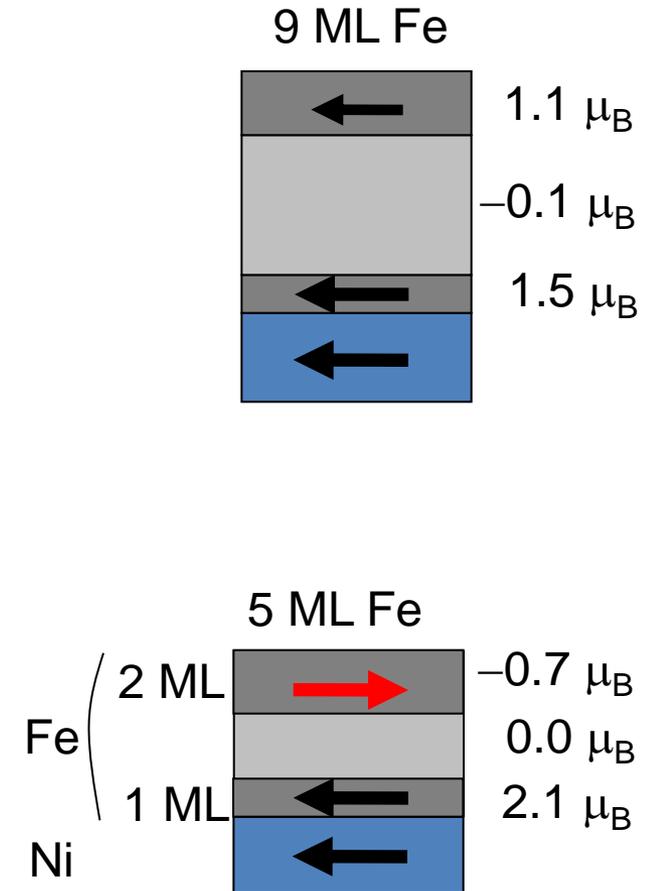
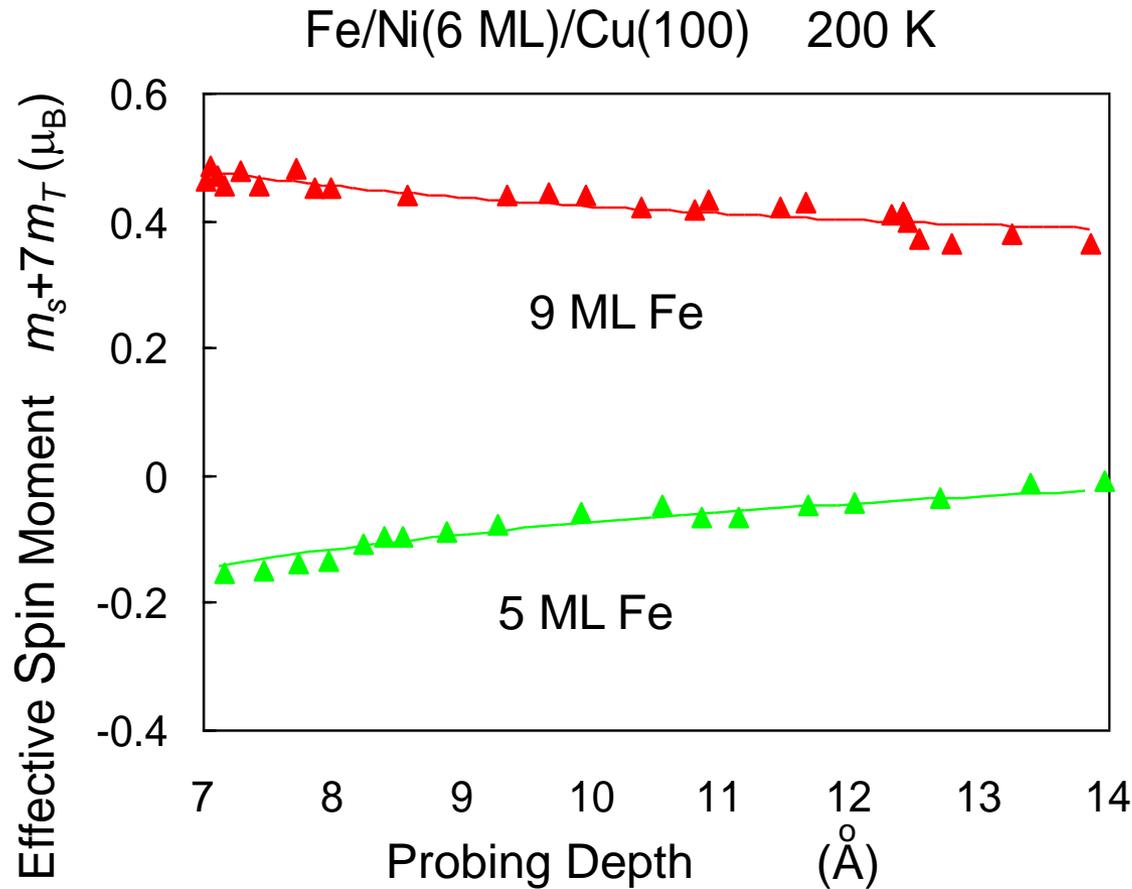
# Fe(x ML)/Ni( 6 ML)/Cu(100)

Fe L-edge Depth-resolved XMCD  
Grazing Incidence (200 K)

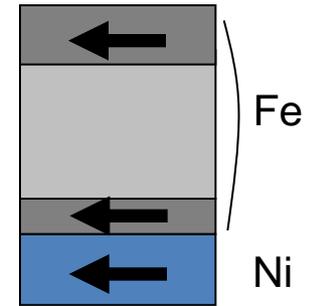
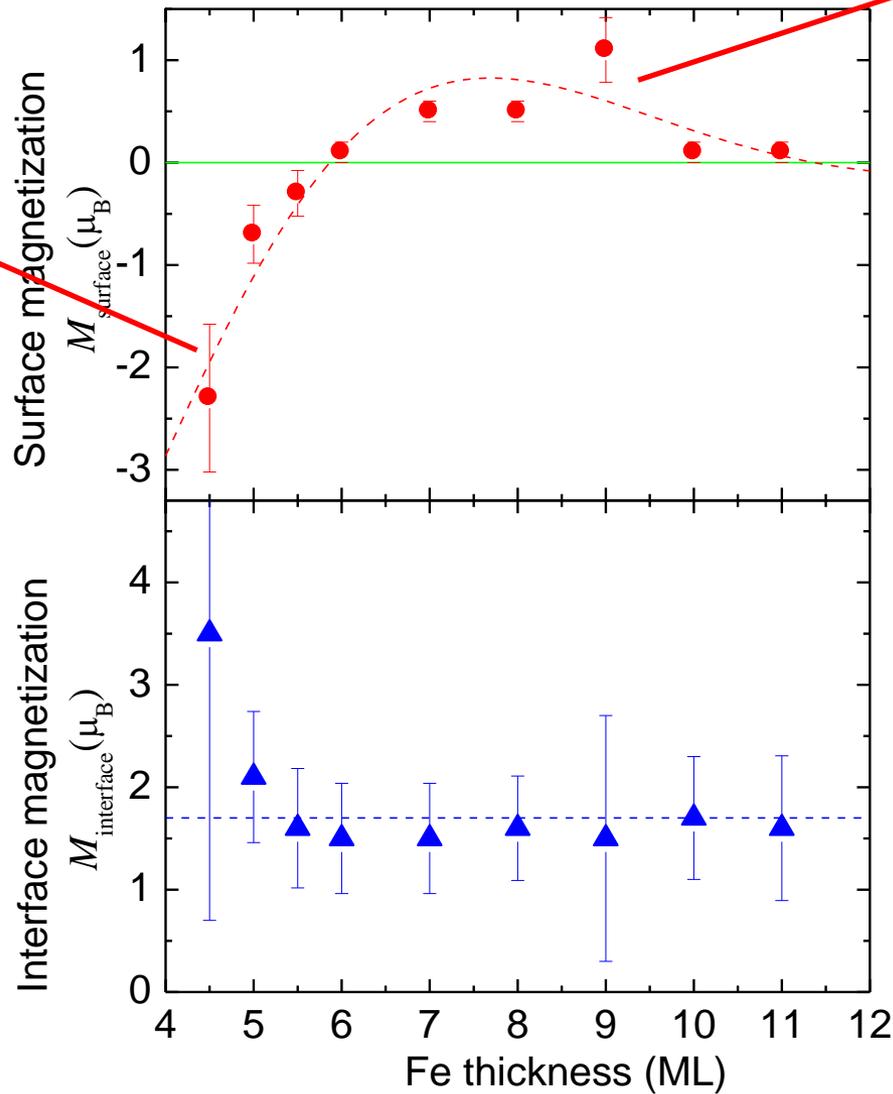
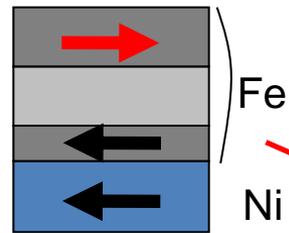




# Curve fitting with a three-regions model



# Fe thickness dependence at 200 K



Oscillatory  
surface magnetization

Positive  
interface magnetization

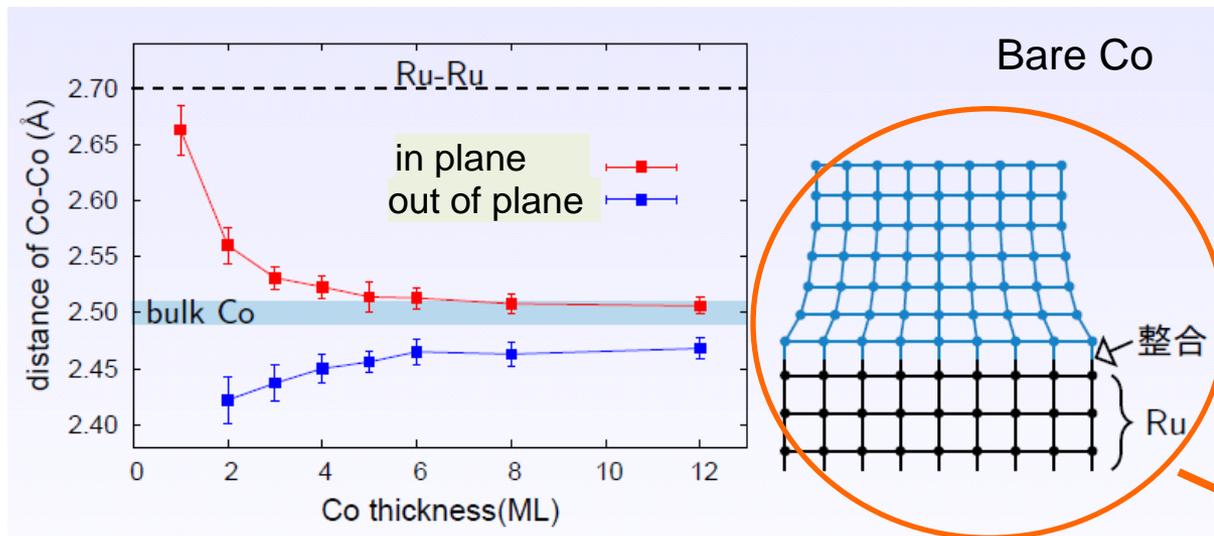


Oscillatory  
magnetic coupling  
between  
surface and interface

# Atomic structure of Ru/Co/Ru(0001) thin films

Fluorescence-yield EXAFS (Co K edge) : average over the whole film

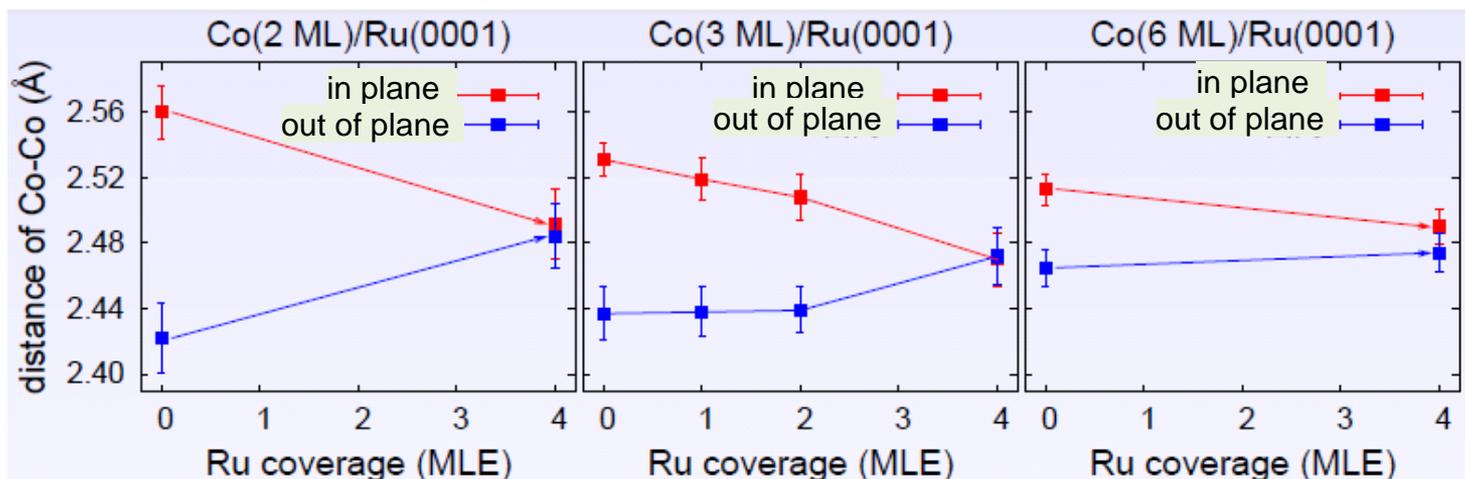
Miyawaki et al.,  
PRB 80 (2009) 020408(R).



Interface Co layer is commensurate to Ru

Rapid relaxation upon further Co deposition

## Effects of Ru capping



Is that true?

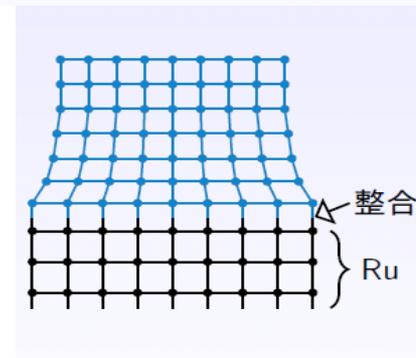
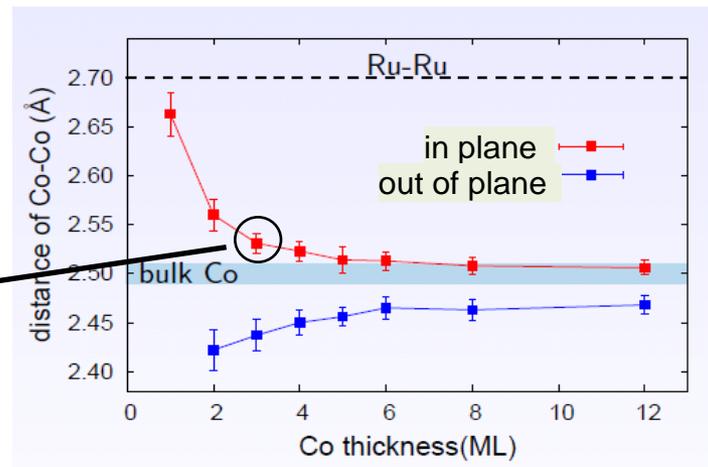
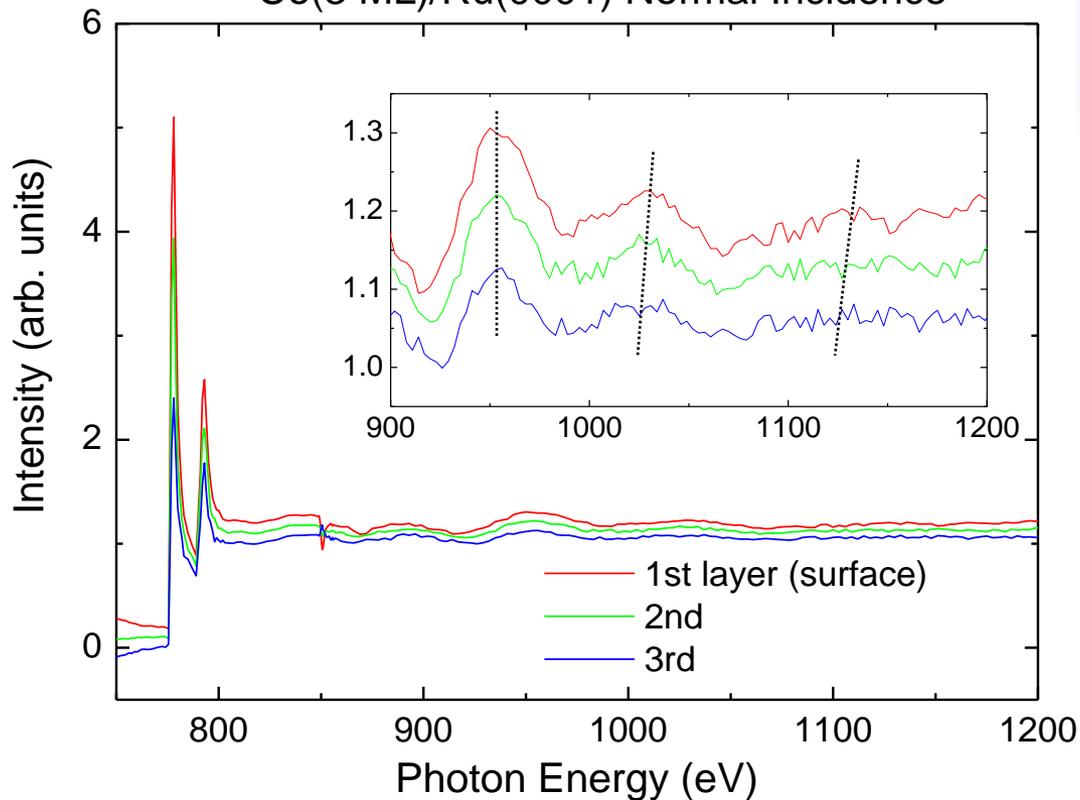
Relaxation of Co distortion upon Ru capping

# Depth profile of atomic structure

Normal incidence: dominated by in-plane distance

## Layer-resolved EXAFS

Co(3 ML)/Ru(0001) Normal Incidence



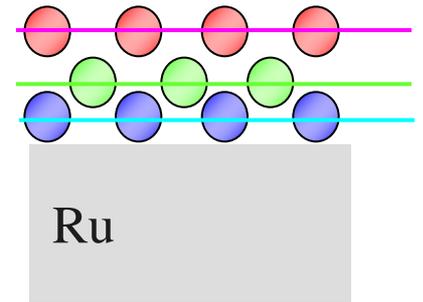
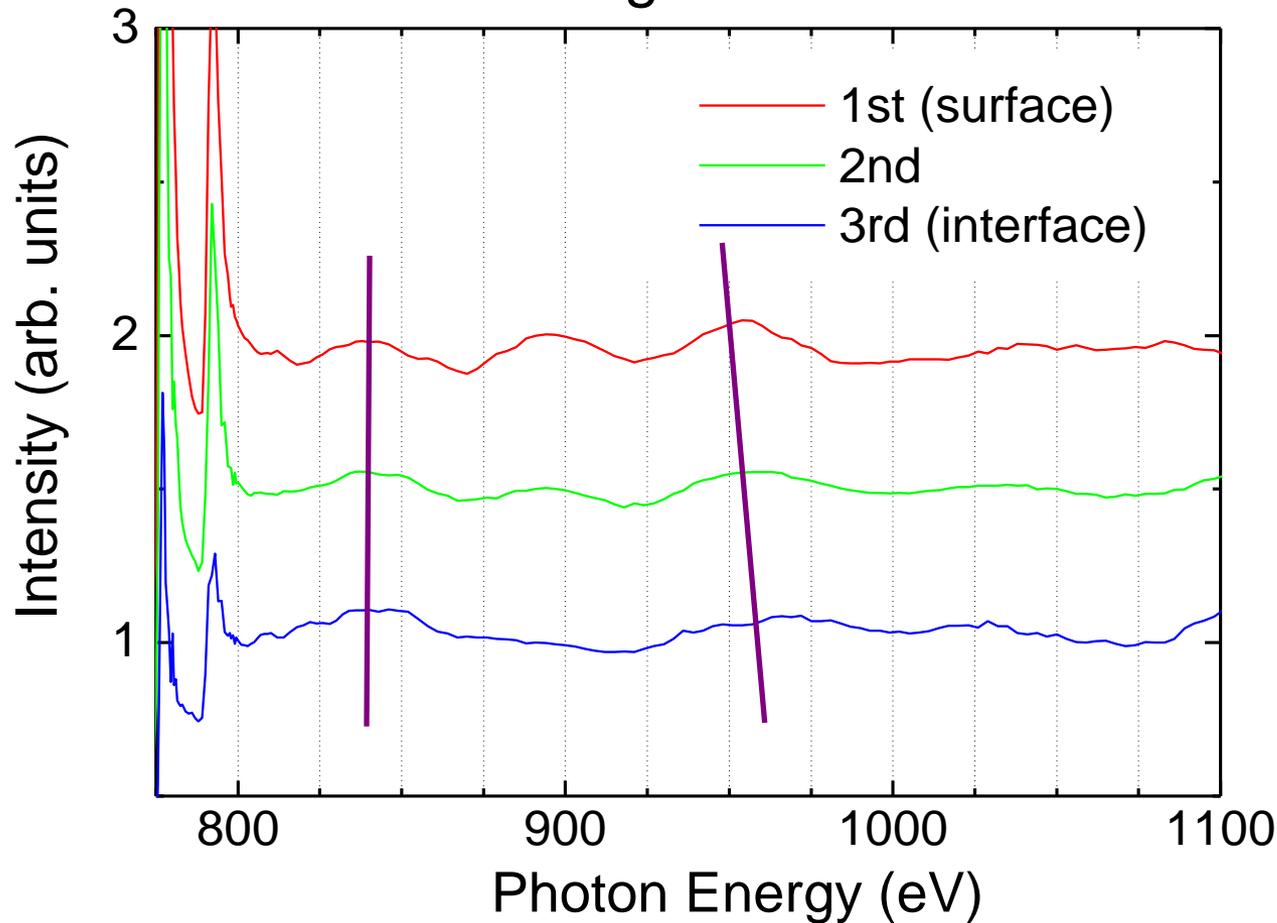
It might be true...

Surface shows longer oscillation period: shorter bond length

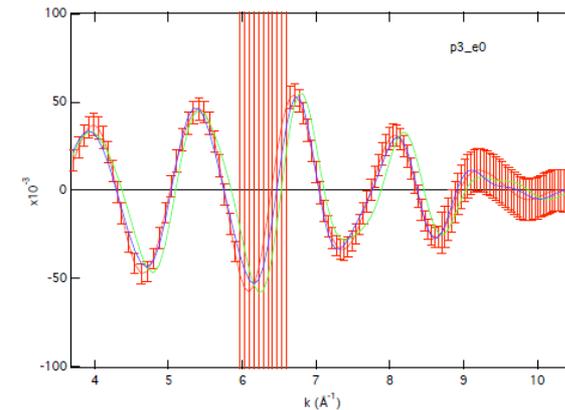
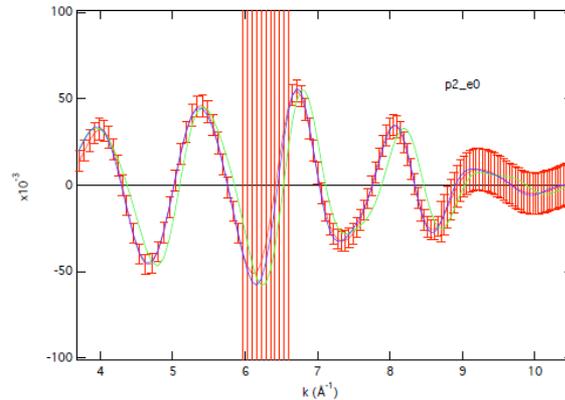
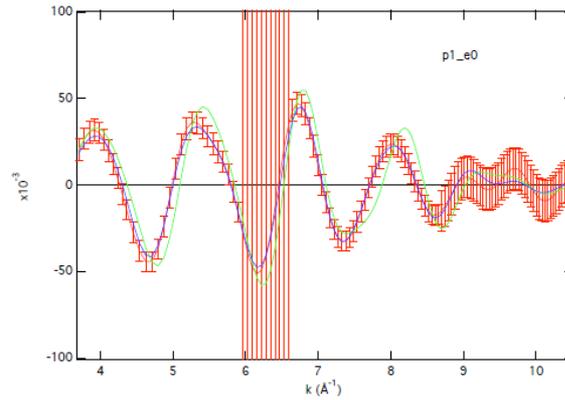
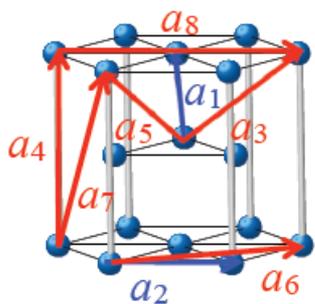
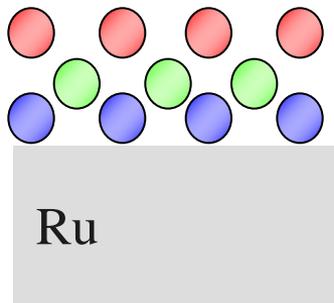
# Depth-resolved EXAFS at grazing incidence

Longer out-of-plane bond length at surface?

## Grazing Incidence



# Preliminary analyses by Bayes-Turchin method



In plane

2.51  $\text{\AA}$

out of plane

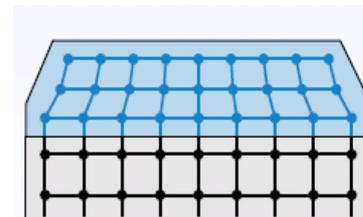
2.60  $\text{\AA}$

2.55  $\text{\AA}$

2.52  $\text{\AA}$

2.55  $\text{\AA}$

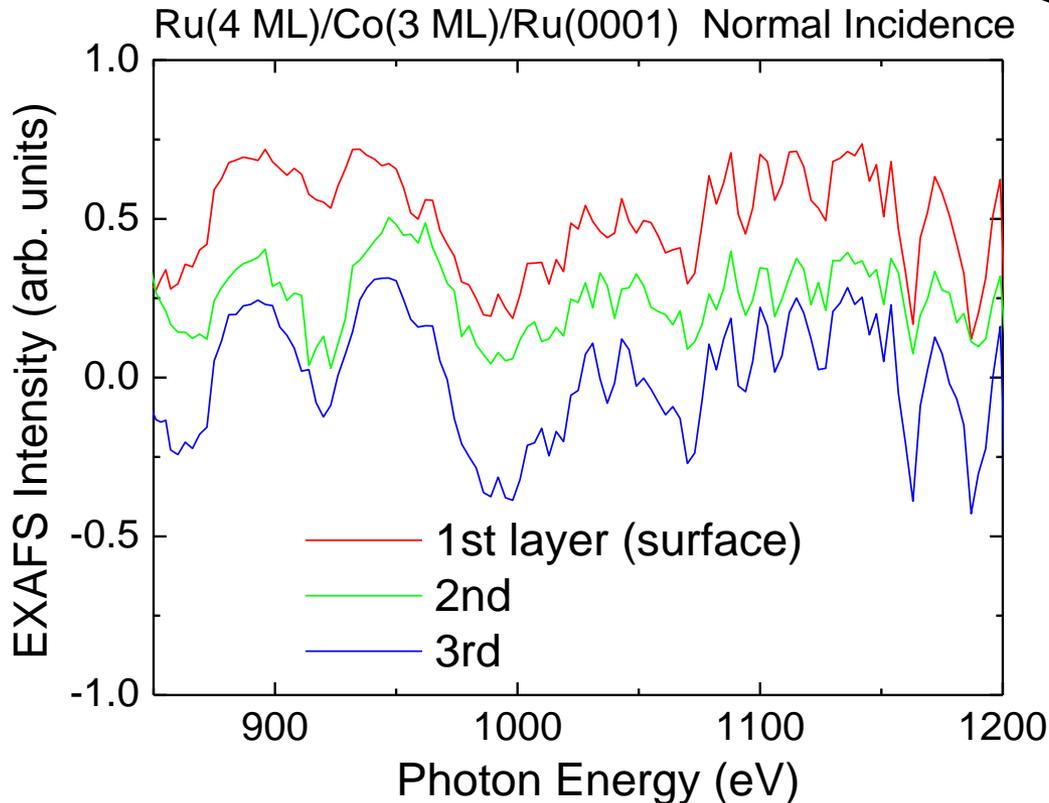
2.52  $\text{\AA}$



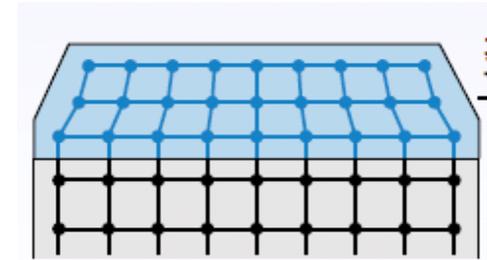
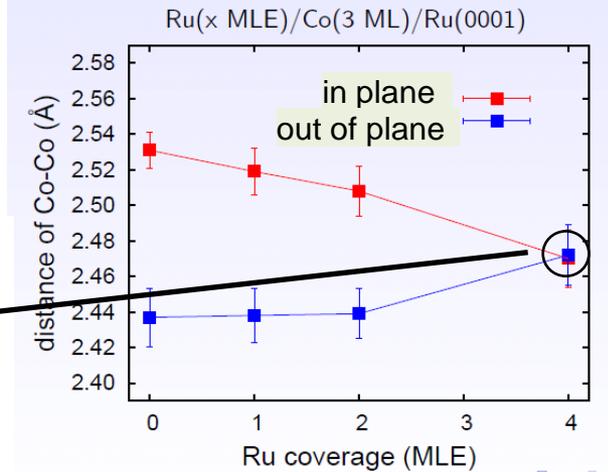
# Effects of Ru capping

Normal incidence: in-plane bond length

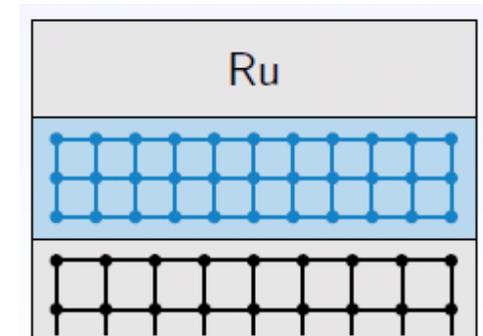
Layer-resolved EXAFS



Little difference in the bond length



↓ Relaxation



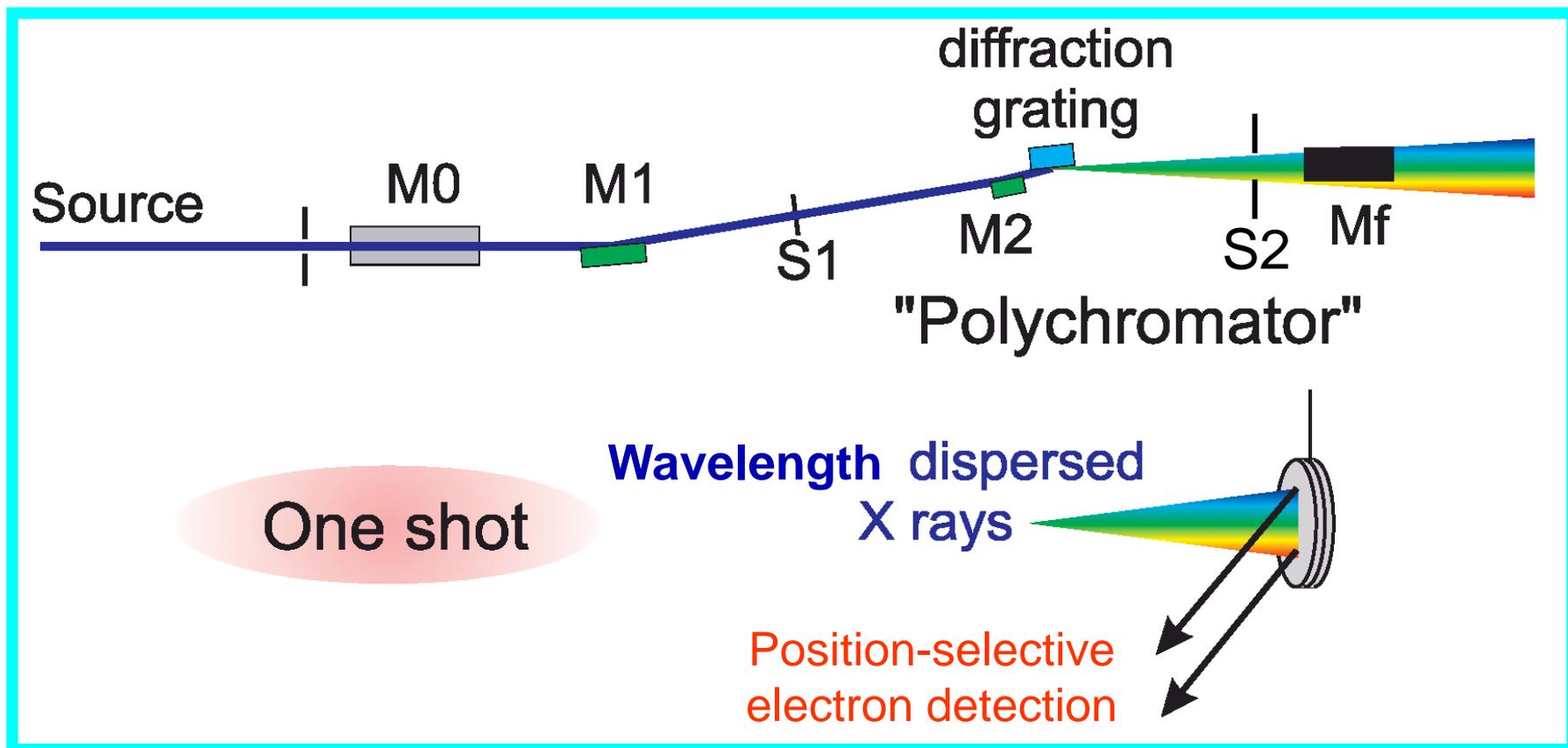
1. Advantages and Disadvantages of  
Soft X-ray Absorption Spectroscopy (SXAS)
2. SXAS studies on Surface and Thin films
3. Novel SXAS Techniques
  - 3-1. Depth-resolved XAS
  - 3-2. Wavelength-dispersive XAS

# Development of Wavelength-dispersive XAS

XAS: **Element selectivity, Chemical species determination, Structural information,...**

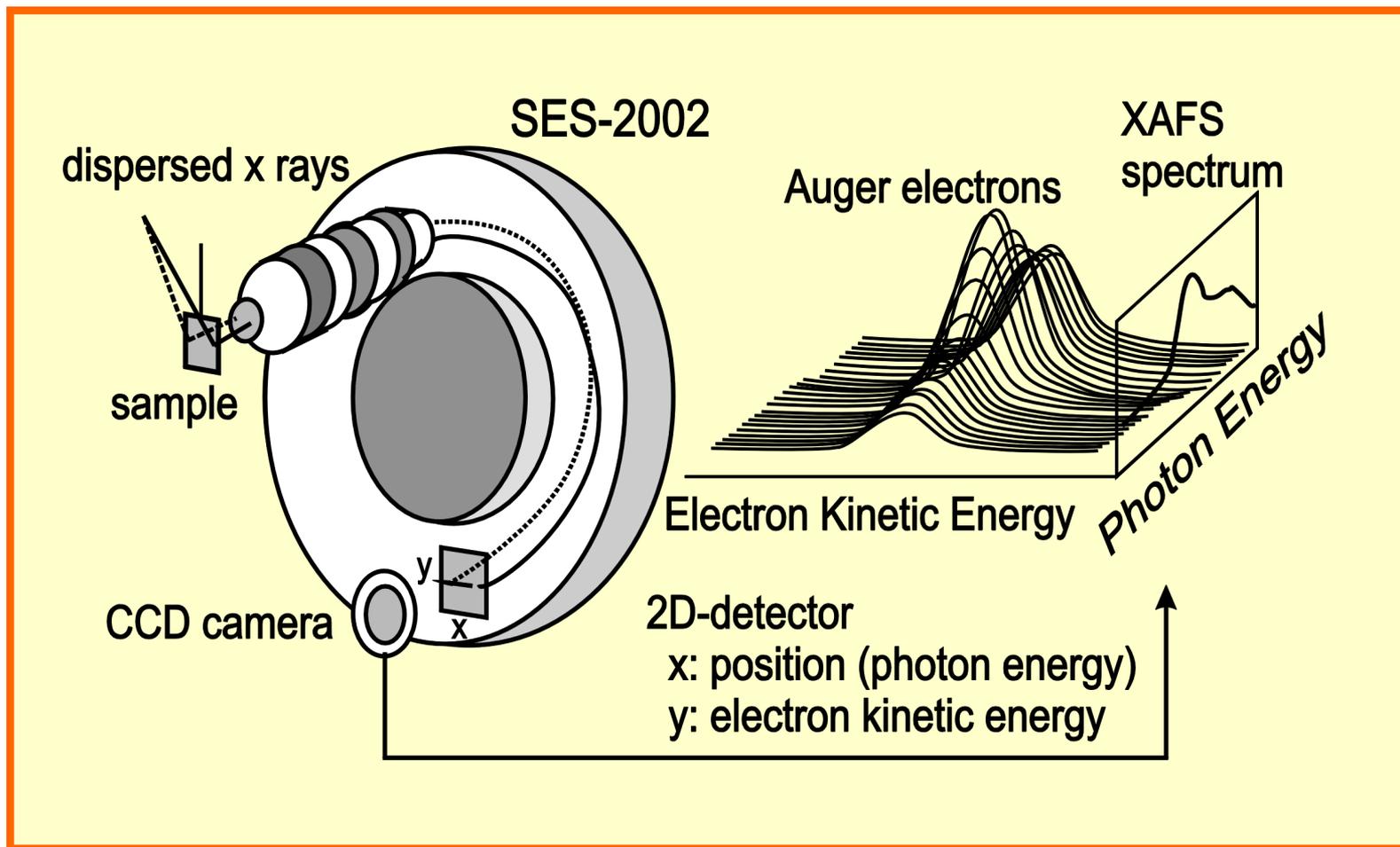
Takes long time (~10 min/spectrum) for measurement

Possibility of "One shot" measurement.



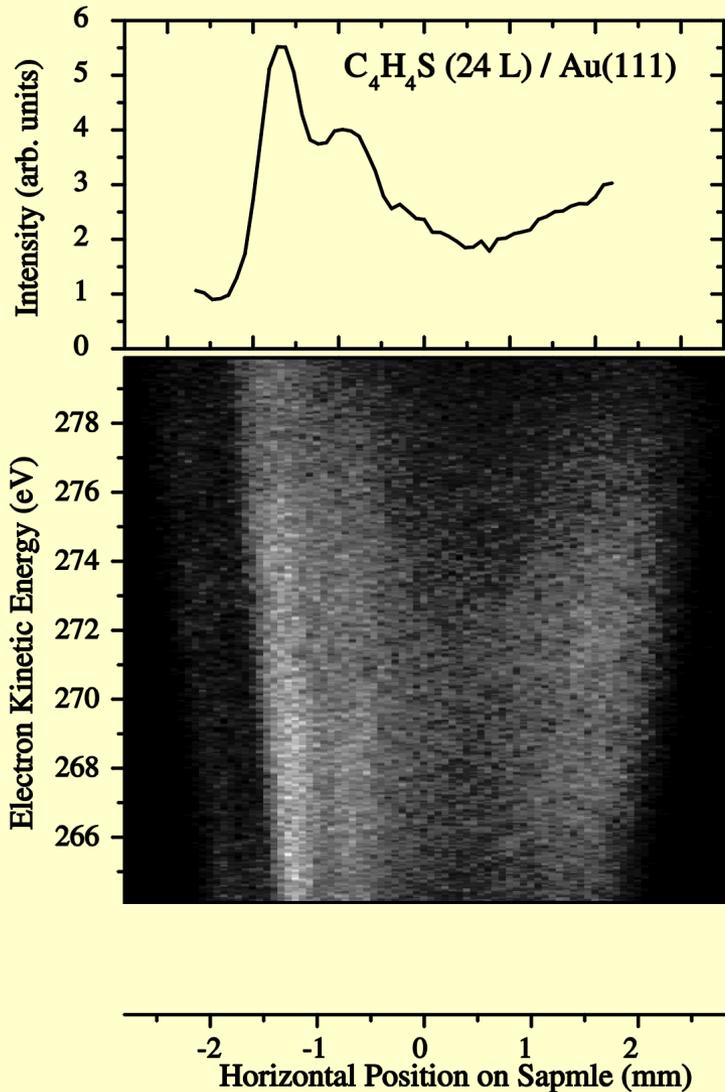
# Experimental setup for wavelength-dispersive XAS

- Wavelength-dispersed X rays + Position-sensitive electron detector

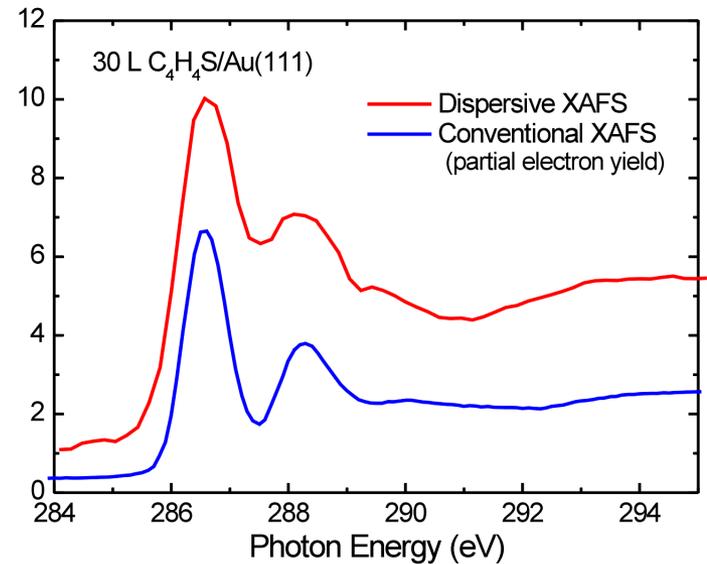


# Test Measurement

Amemiya et al., Jpn. J. Appl. Phys. **40**, (2001) L718.

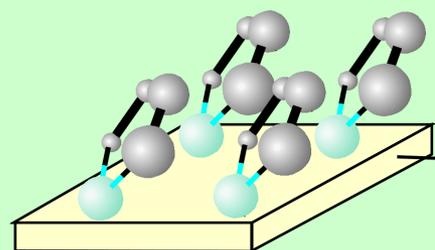


## Comparison with conventional XAS

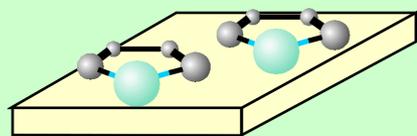


~100 times faster!

# Example: $C_4H_4S/Au(111)$



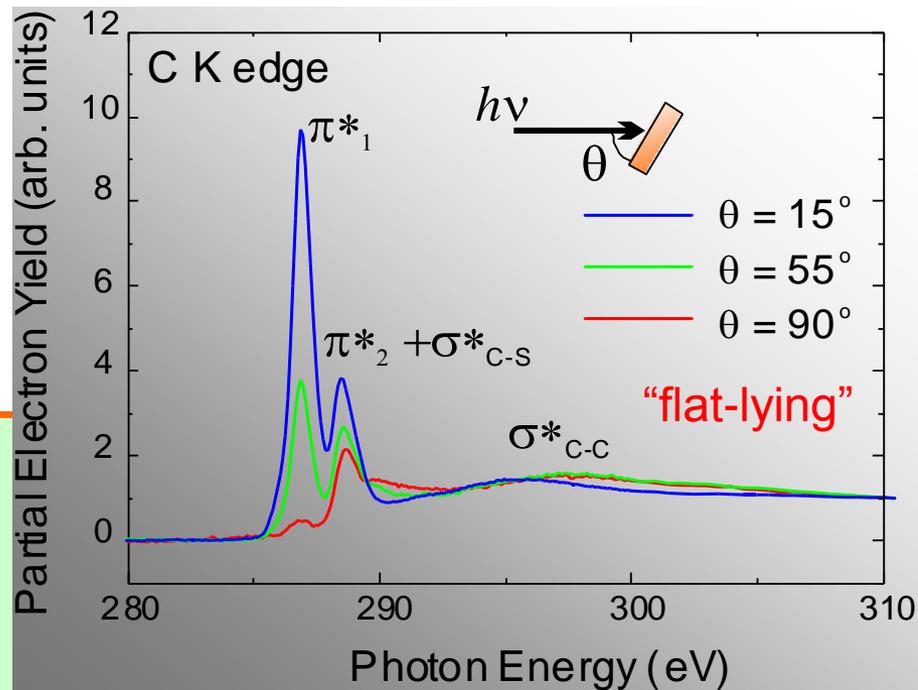
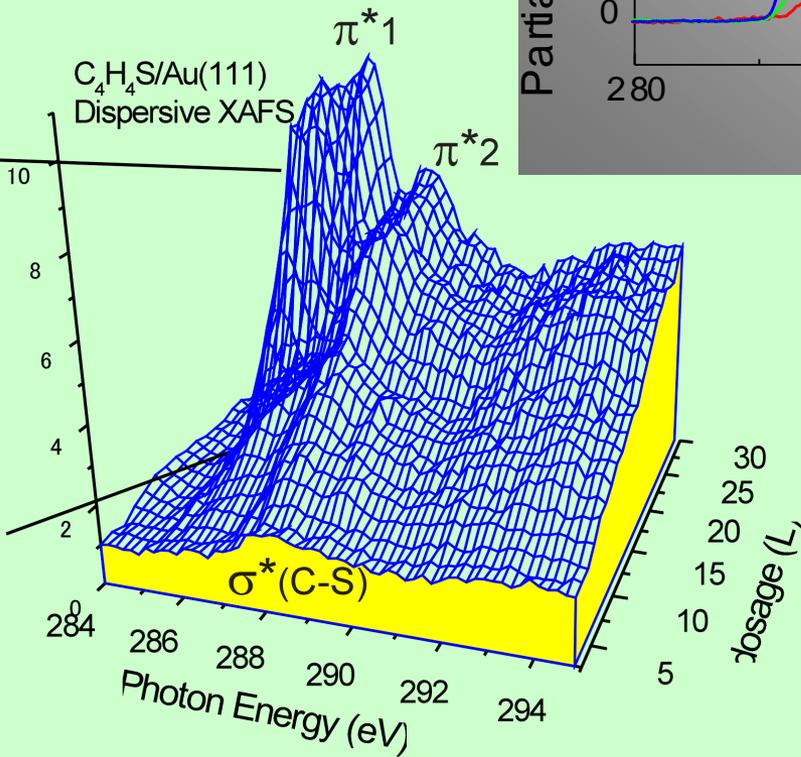
tilted



flat-lying

$\theta = 90^\circ$

$C_4H_4S/Au(111)$   
Dispersive XAFS

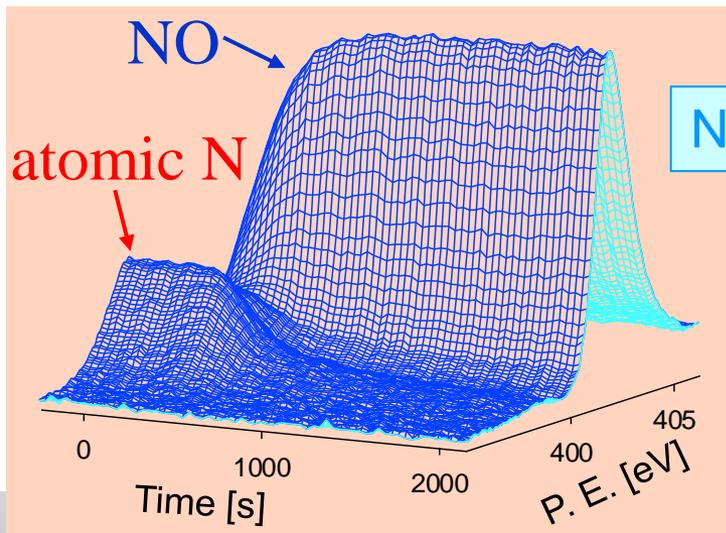


Orientation change  
with increasing coverage

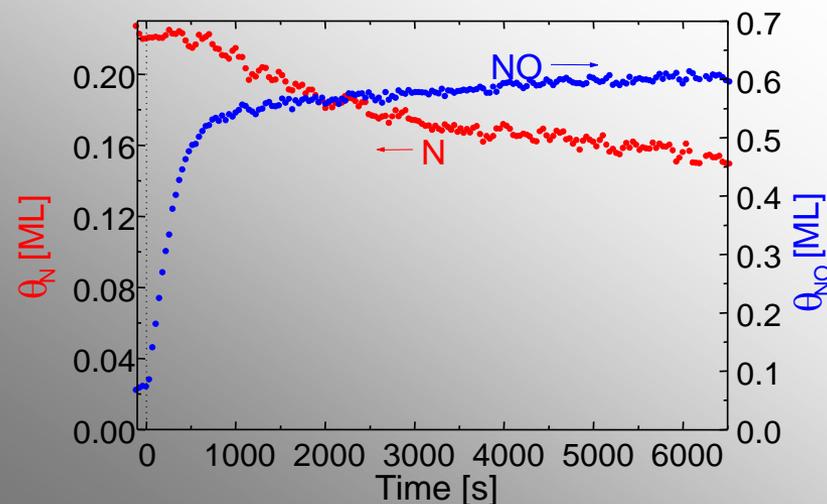
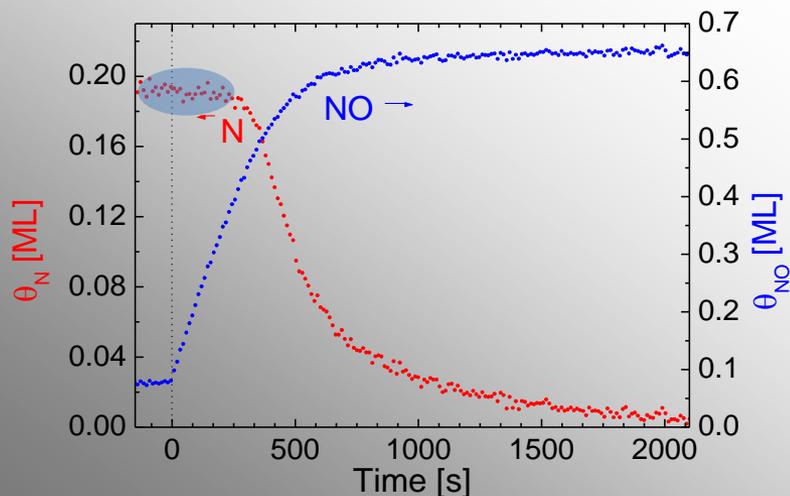
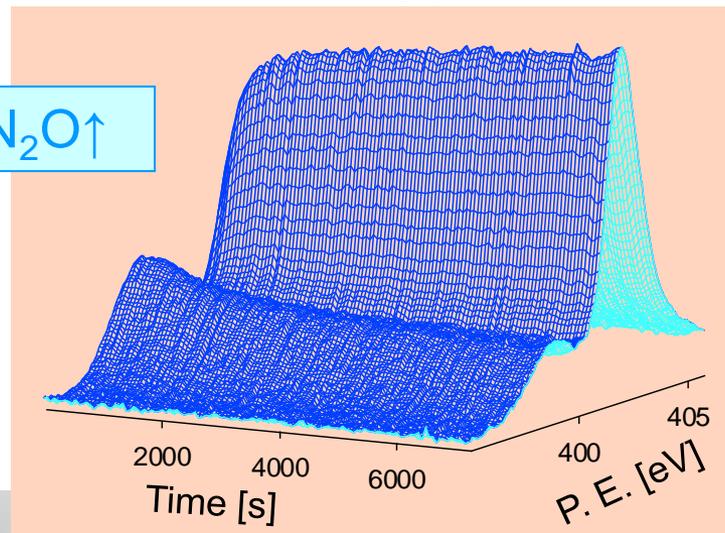
# Chemical Reaction: NO/N/Rh(111)

Nakai et al., J. Phys. Chem. B **110** (2006) 25578.

T=120 K



T=250 K



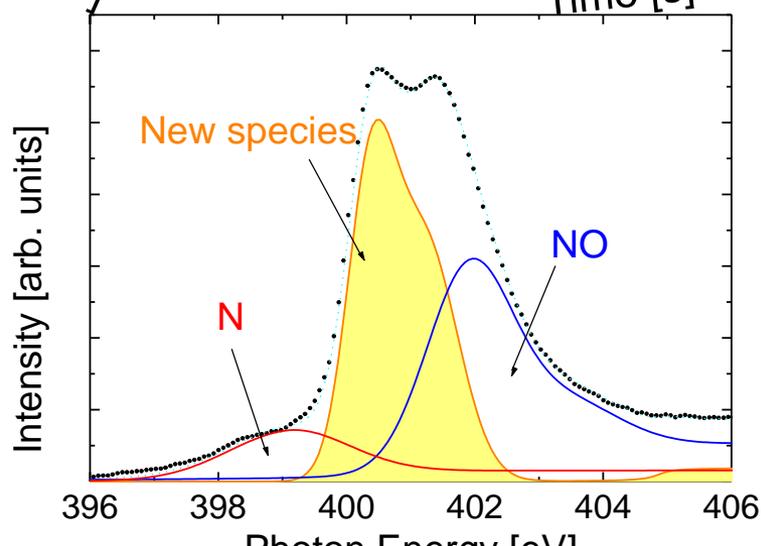
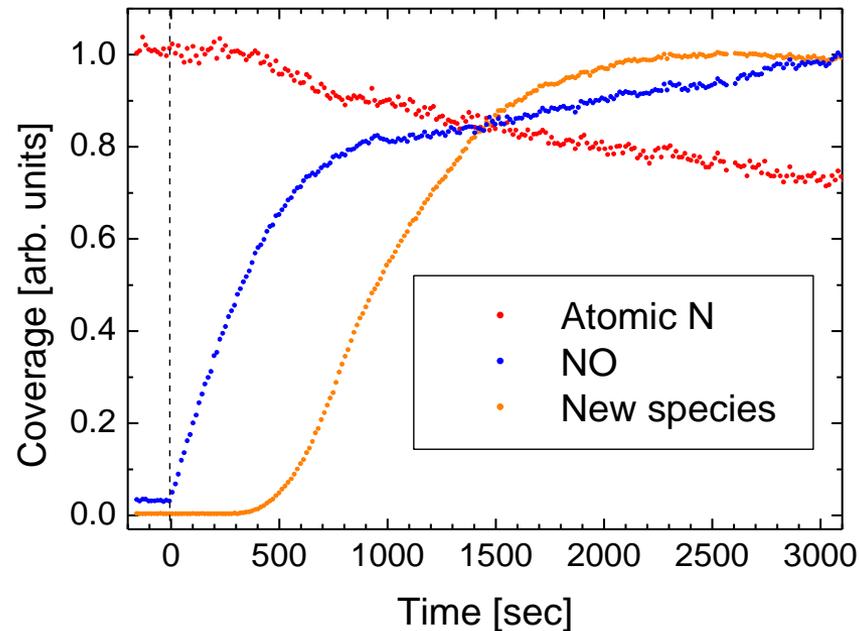
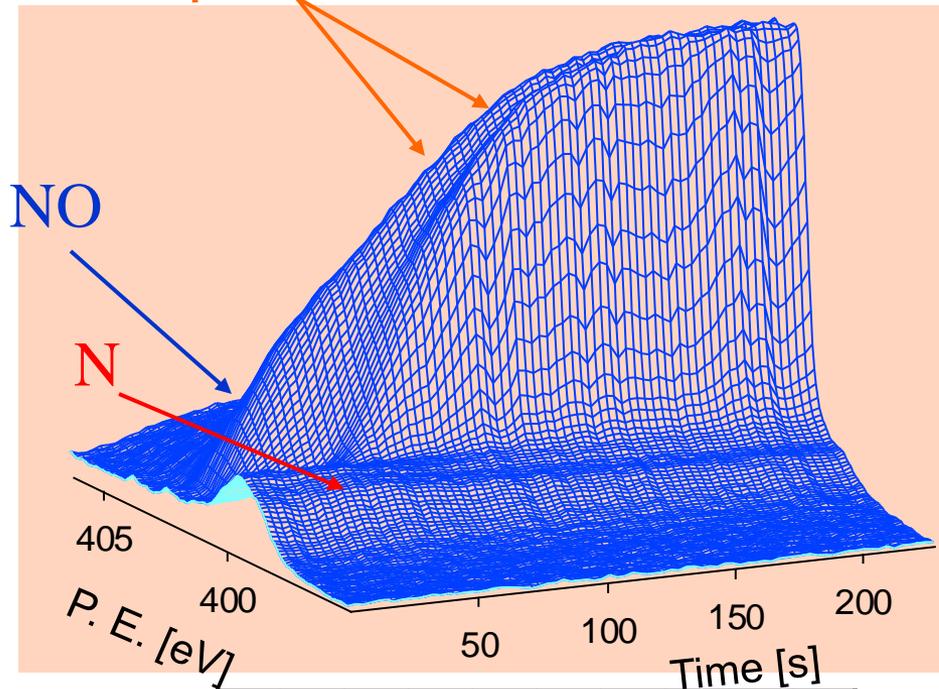
- Induction period: reaction does not start immediately
- Faster reaction at lower temperature

# Lower Temperature

$T=70\text{ K}$ ,  $P_{\text{NO}}=5\times 10^{-9}\text{ Torr}$

Nakai et al., J. Phys. Chem. B 126 (2007) 044704.

## New species



Appearance of new species

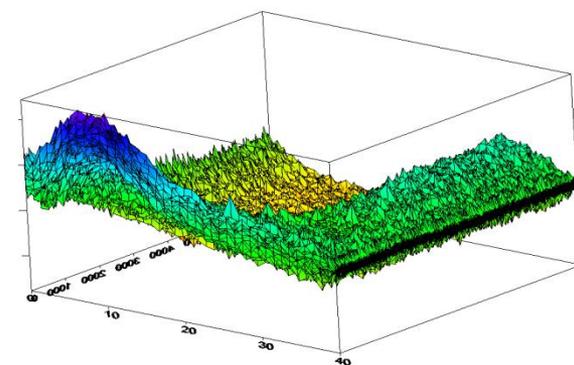
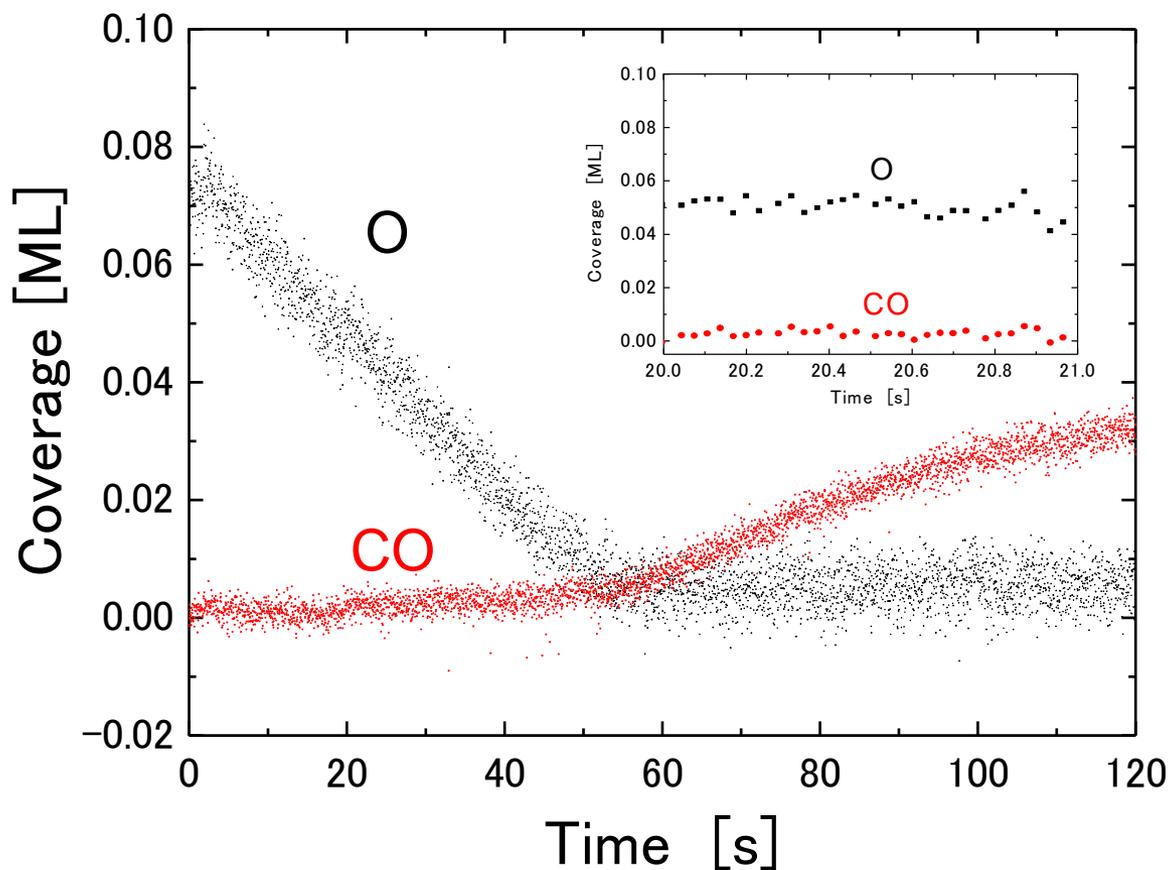
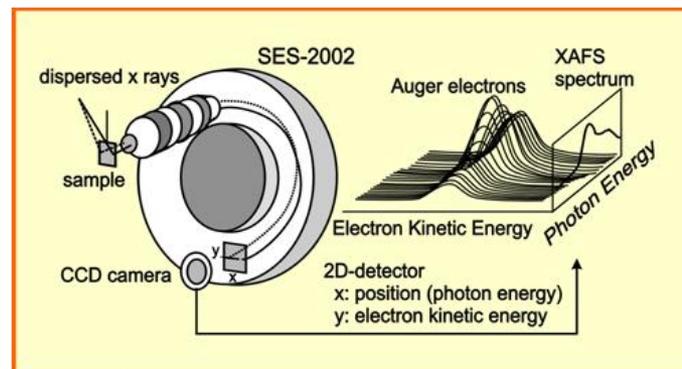
$\Rightarrow$  NO dimer

“New species” might be precursor.

# Further Development

Undulator beamline (BL-16A)

Video rate (~30 Hz)



400 K

Experimental conditions

O : p(2x2)

CO:  $P_{\text{CO}} = 1 \times 10^{-8}$  Torr

acquisition time:

33 ms/spectrum

**Thank you for your attention!**